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# Advances In 3D Tomography And 4DSTEM: Perspectives To Study Semiconductor Devices

Nicolas Gauquelin, Daniel Arenas Esteban, Mikhail Mychinko, Andrey Orekhov, Daen Jannis, Evgenii Vlasov, Sandra van Aert, Johan Verbeeck and Sara Bals

EMAT, University of Antwerpen, Antwerpen, Belgium

# Outline

- Introduction
  - Semiconductors metrology and its challenges
- Advances in tomography of nanoparticles
  - Electron tomography introduction
  - Conventional vs. Fast tomography
  - Atomic resolution tomography
  - SEEBIC as an alternative
- 4DSTEM in materials science
  - Introduction to 4DSTEM: Orientation mapping, stacking determination and twist angle in the TEM
  - 4DSTEM in SEM on 2D Materials
  - Fluctuation microscopy in ZrCu metallic glasses
  - Biasing -  $\text{Ca}_2\text{RuO}_4$ : Stripes formation after quench of the electric field
- Multimodal tomography on a semiconductor device
  - Introduction to GAA-FET
  - HAADF tomography
  - EELS-EDX tomography
  - 4DSTEM
- Perspectives and outlook



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# Increasing complexity of semiconductor technology

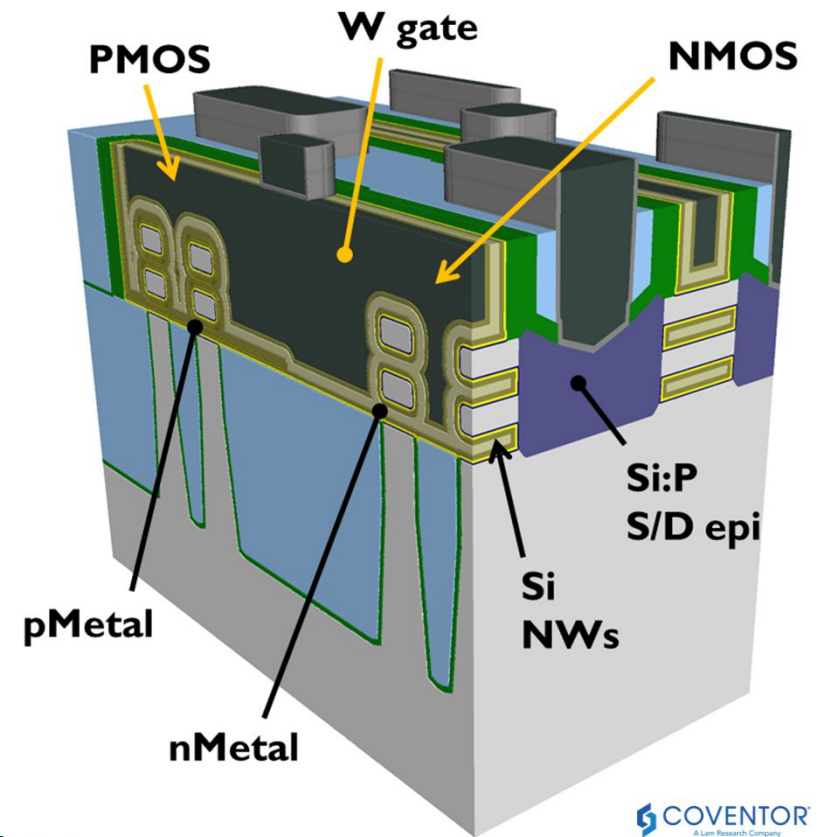


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From IMEC

# Challenges for TEM analysis

- Interface sharpness
- Ge diffusion
- Composition
- Layer continuity
- Shape and size
- Strain evolution
- 3D device
- Nanowire dimensions < typical TEM specimen thickness
- Many materials in nm layer thickness



Courtesy of P. Favia - IMEC

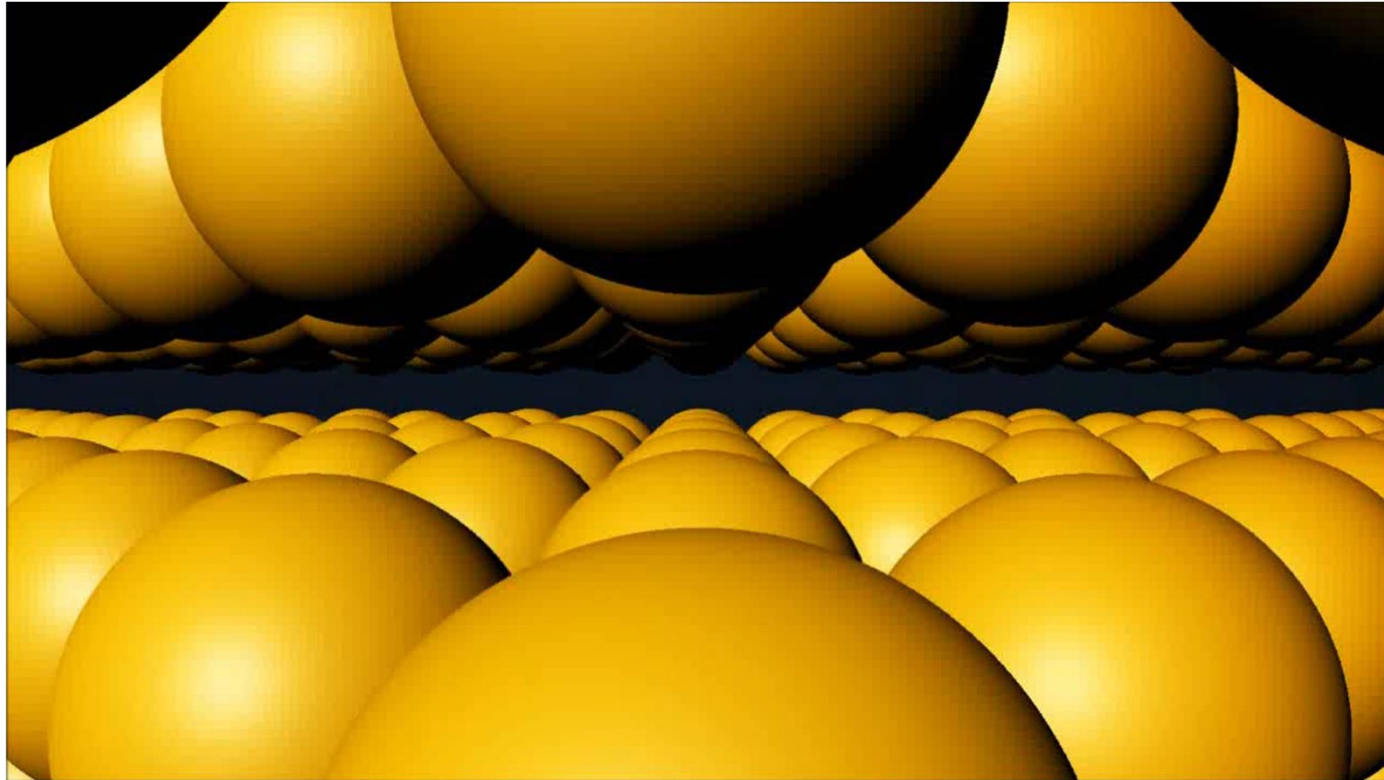
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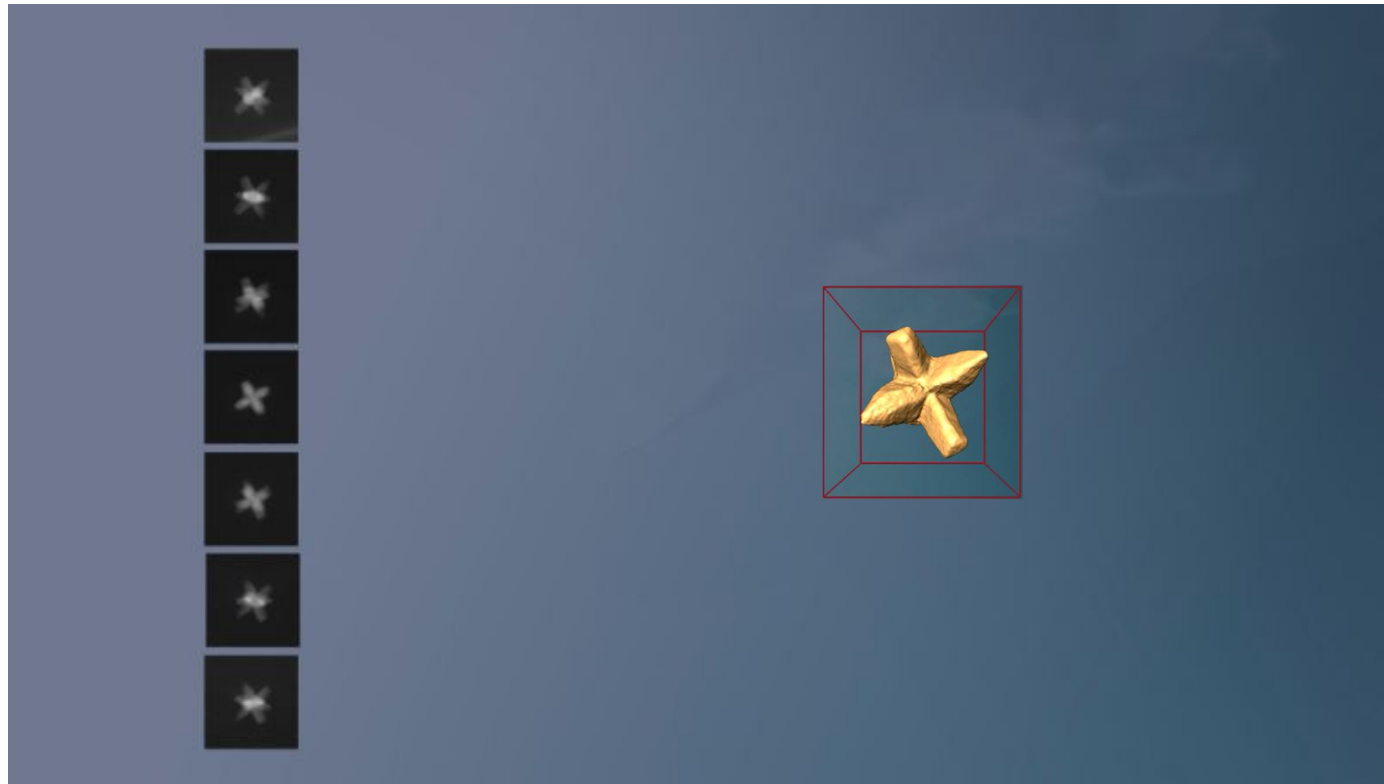
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# In the electron microscope



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# From the 2D images to the 3D object



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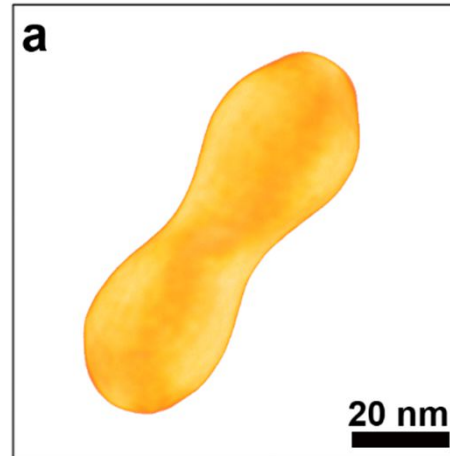
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# Fast tomography

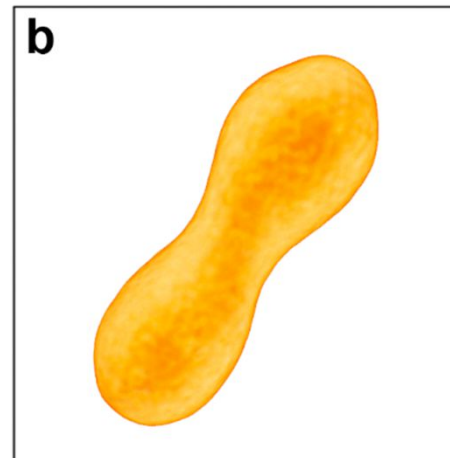
Normal  
Tomo



Fast  
Tomo



49 images  
1 hour



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W. Albrecht and S. Bals, J. Phys. Chem. C 2020, 124, 50, 27276–27286 (2020)



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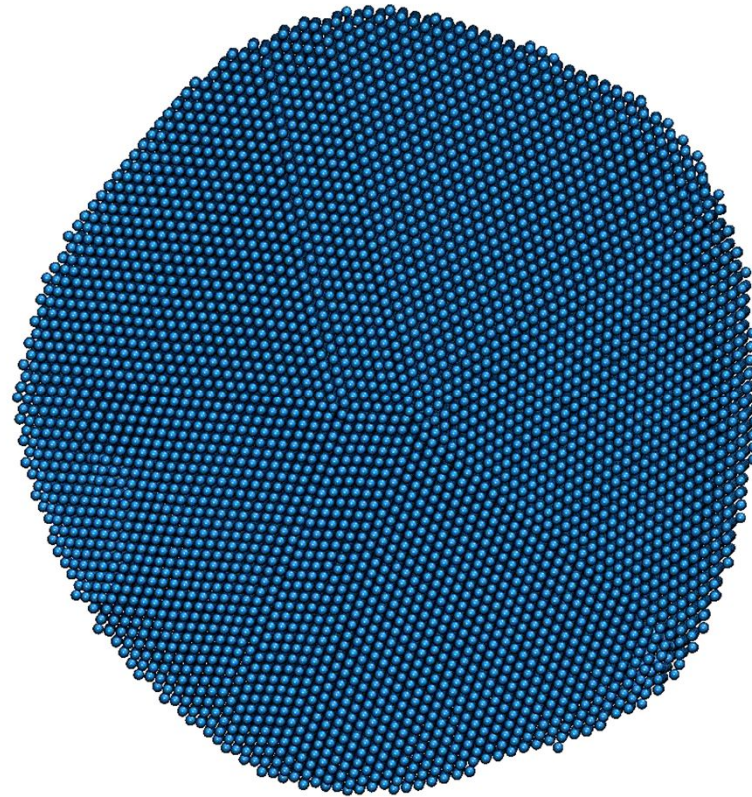
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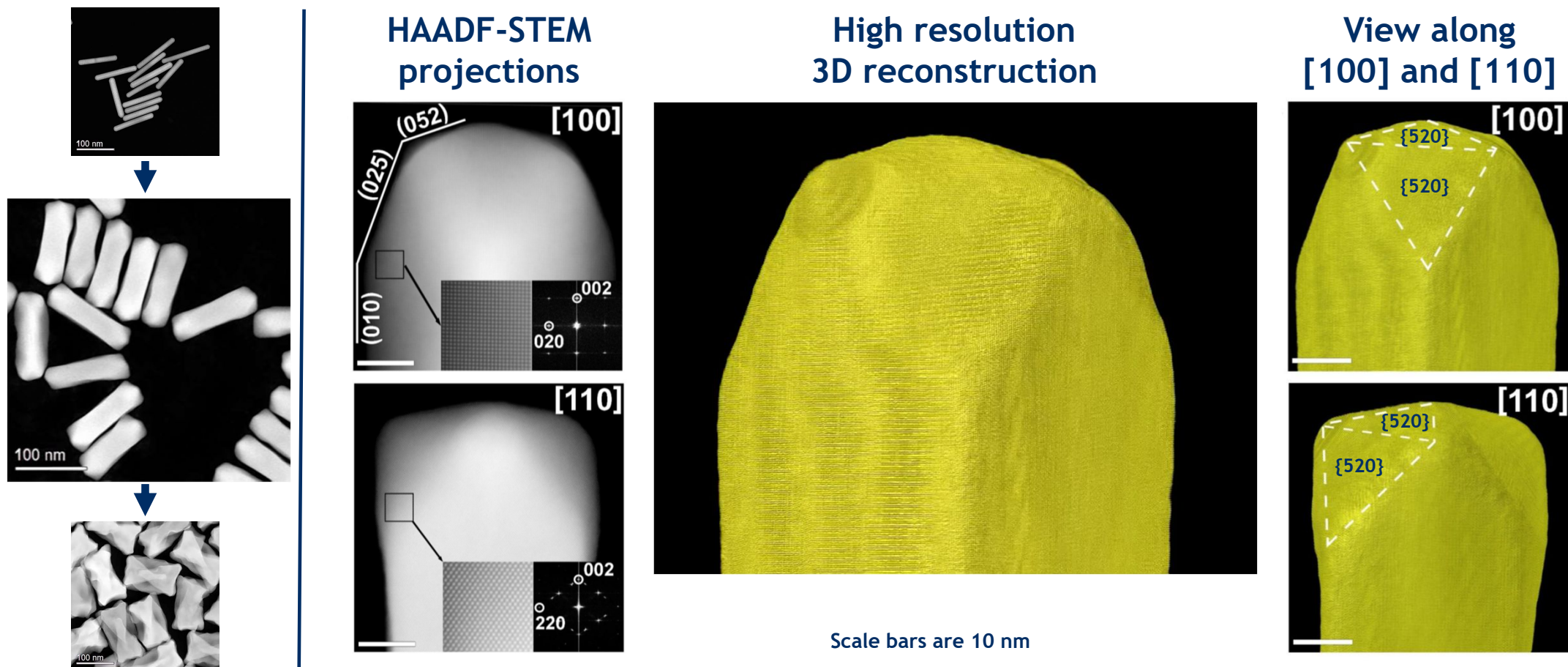
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# Atomic tomography



# Chiral structure and facets



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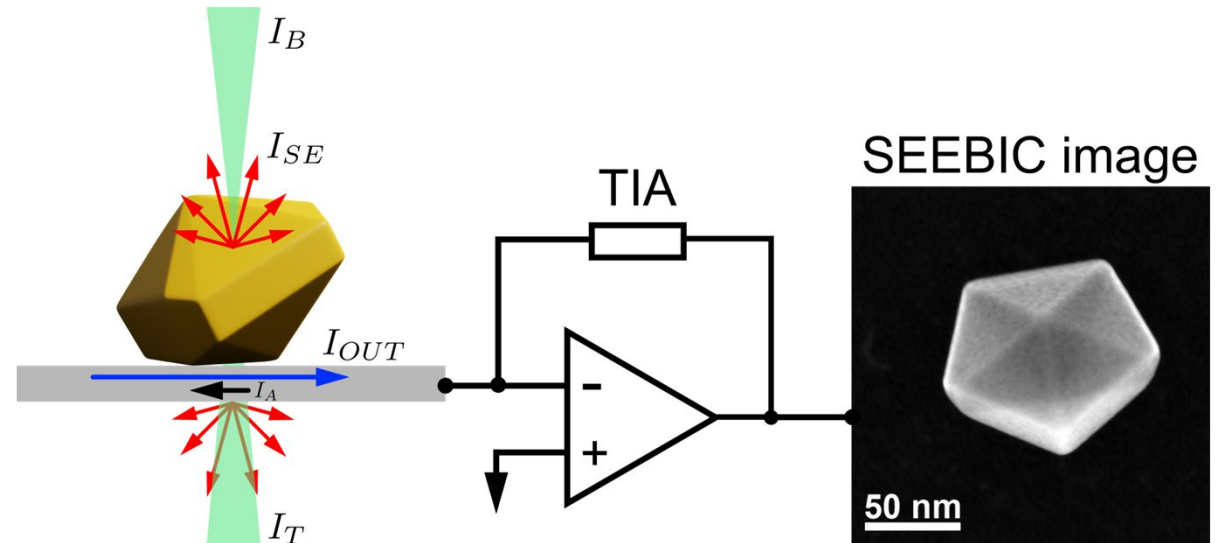


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# SEEBIC principles

## SEEBIC

- signal arises from holes induced by the emission of SEs from the electron-transparent sample
- detected signal is equal but opposite to the generated SE current

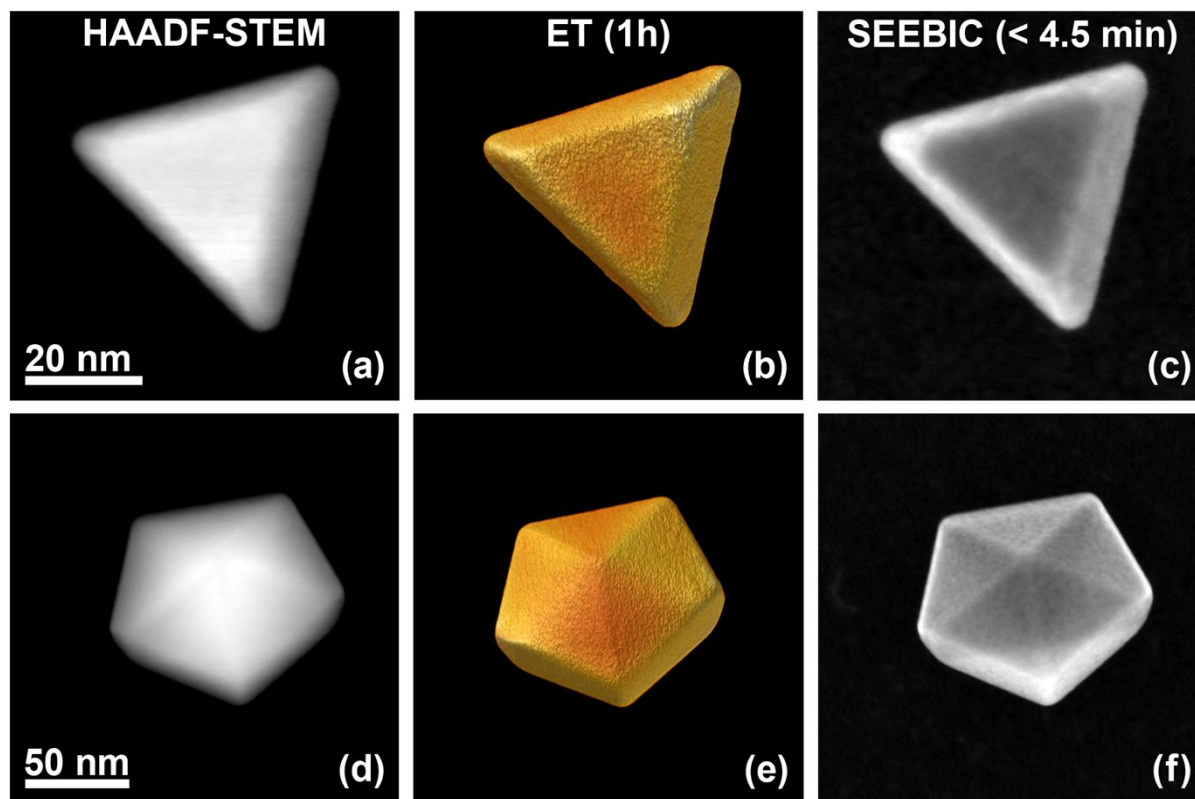


$$I_B = I_T + I_A + I_{BSE} + I_{SE} + V_S/R_S + I_{out}$$

$$-I_{OUT} = I_{SE} + V_S/R_S$$

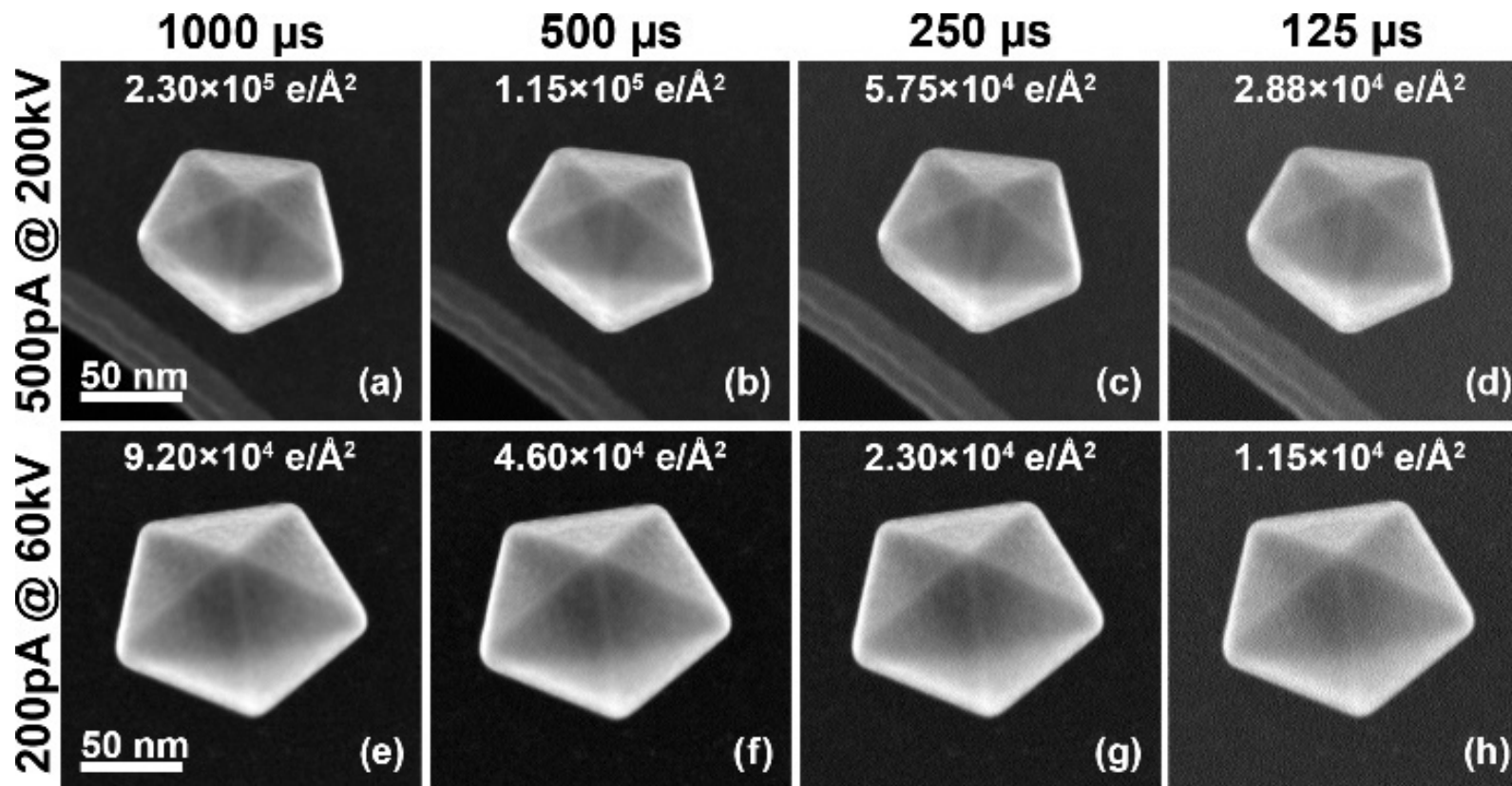


# SEEBIC for visualization of NP morphology



- SEEBIC enables to gain information about surface morphology from a single image in several minutes, whereas ET may take up to 1 hour of acquisition time for tilt series and requires additional processing

# SEEBIC for visualization of NP morphology



>100× faster



2× more dose efficient



4.5 min



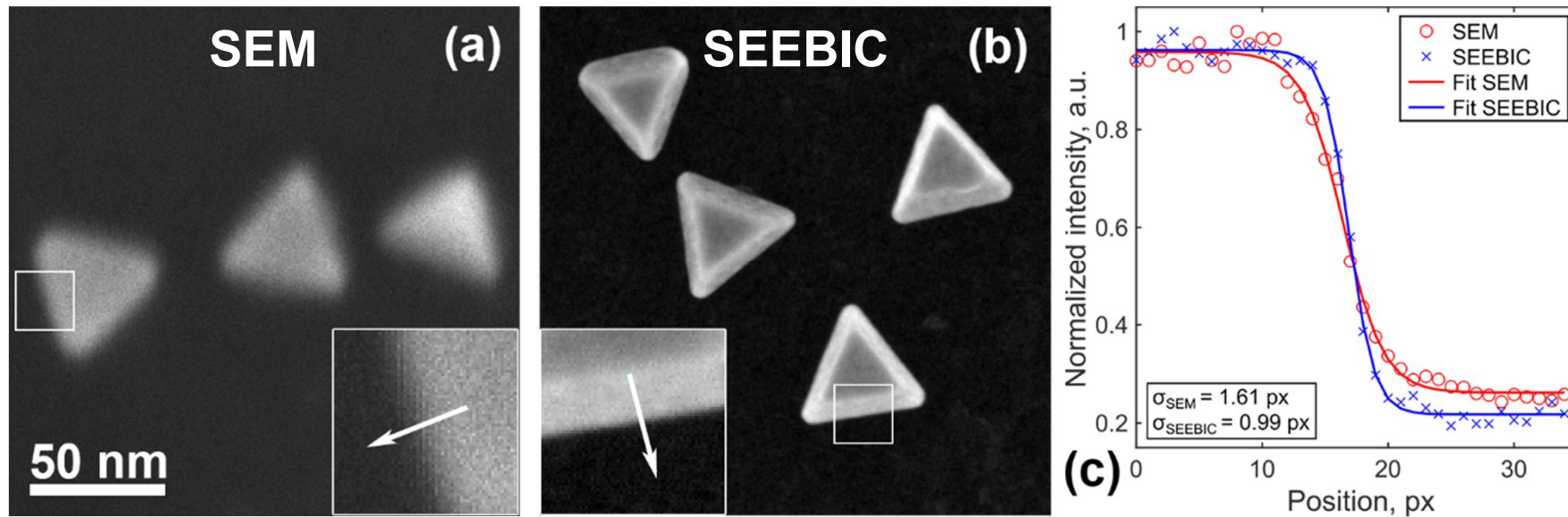
0.5 min



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E. Vlasov et al., *ACS Mater. Lett.*, **5** (2023)

# SEEBIC vs SEM



E. Vlasov, A. Skorikov, A. Sánchez-Iglesias, L. M. Liz-Marzán, J. Verbeeck, S. Bals, *ACS Mater. Lett.*, **5** (2023)

- SEEBIC has superior spatial resolution compared to SEM (1.3 nm vs 4.2 nm)
- SEEBIC resolution is limited by the selected sampling and can be pushed to the obtainable probe size ( $\sim 100 \text{ pm}$ )

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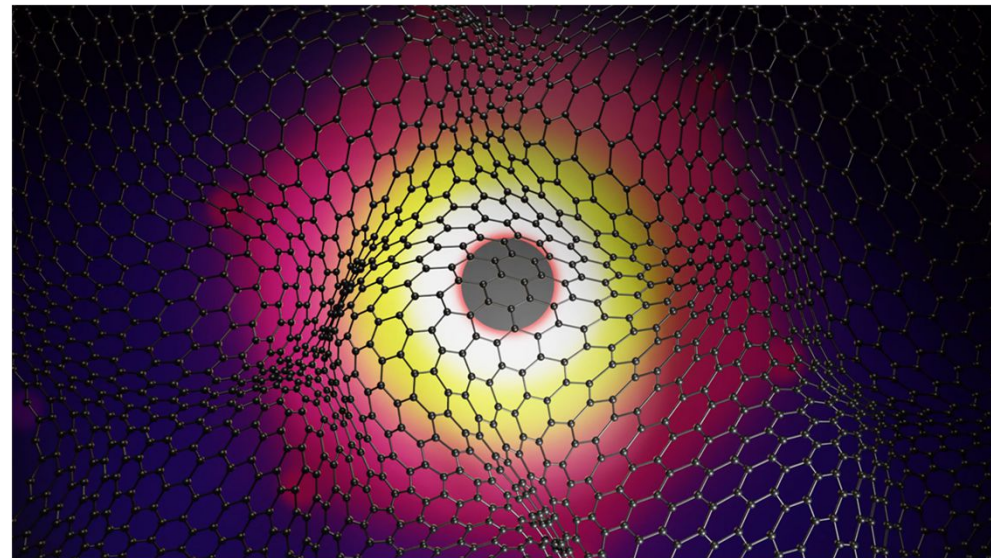
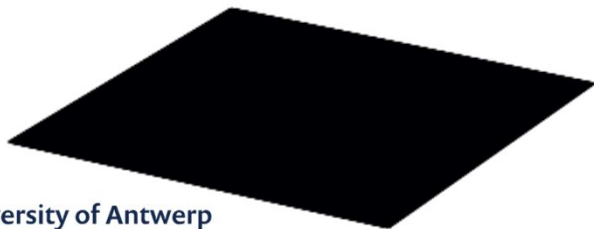
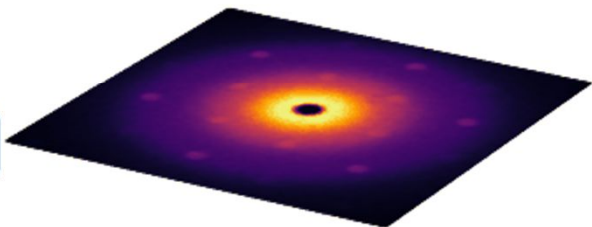
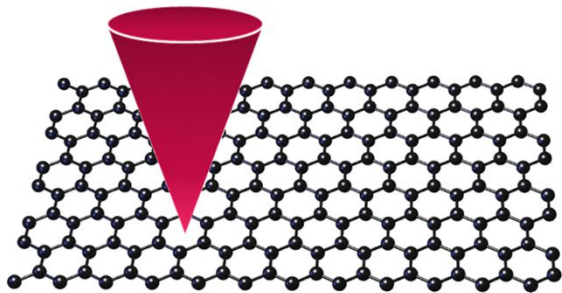
# 4D STEM

Electron  
beam

Specimen

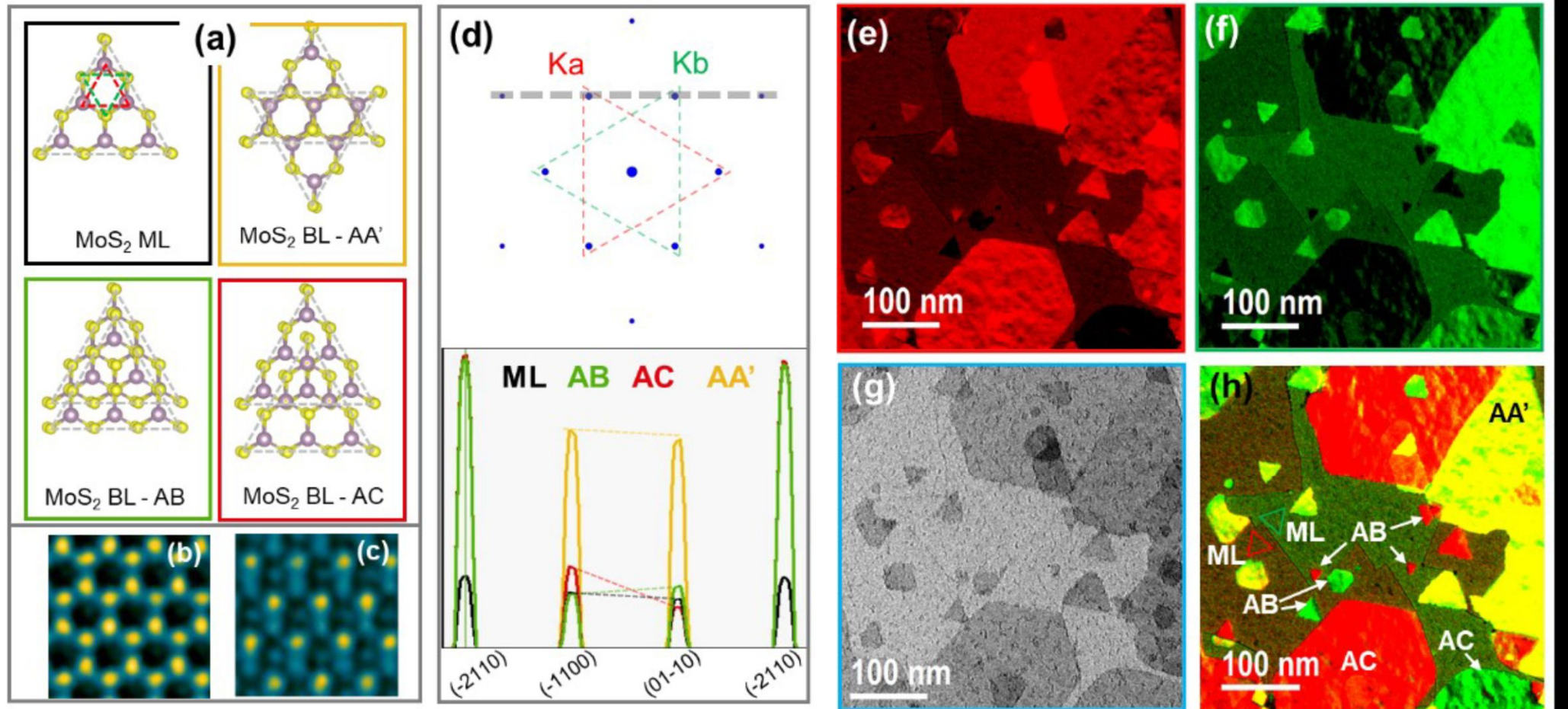
Pixelated  
Detector

Virtual  
Image

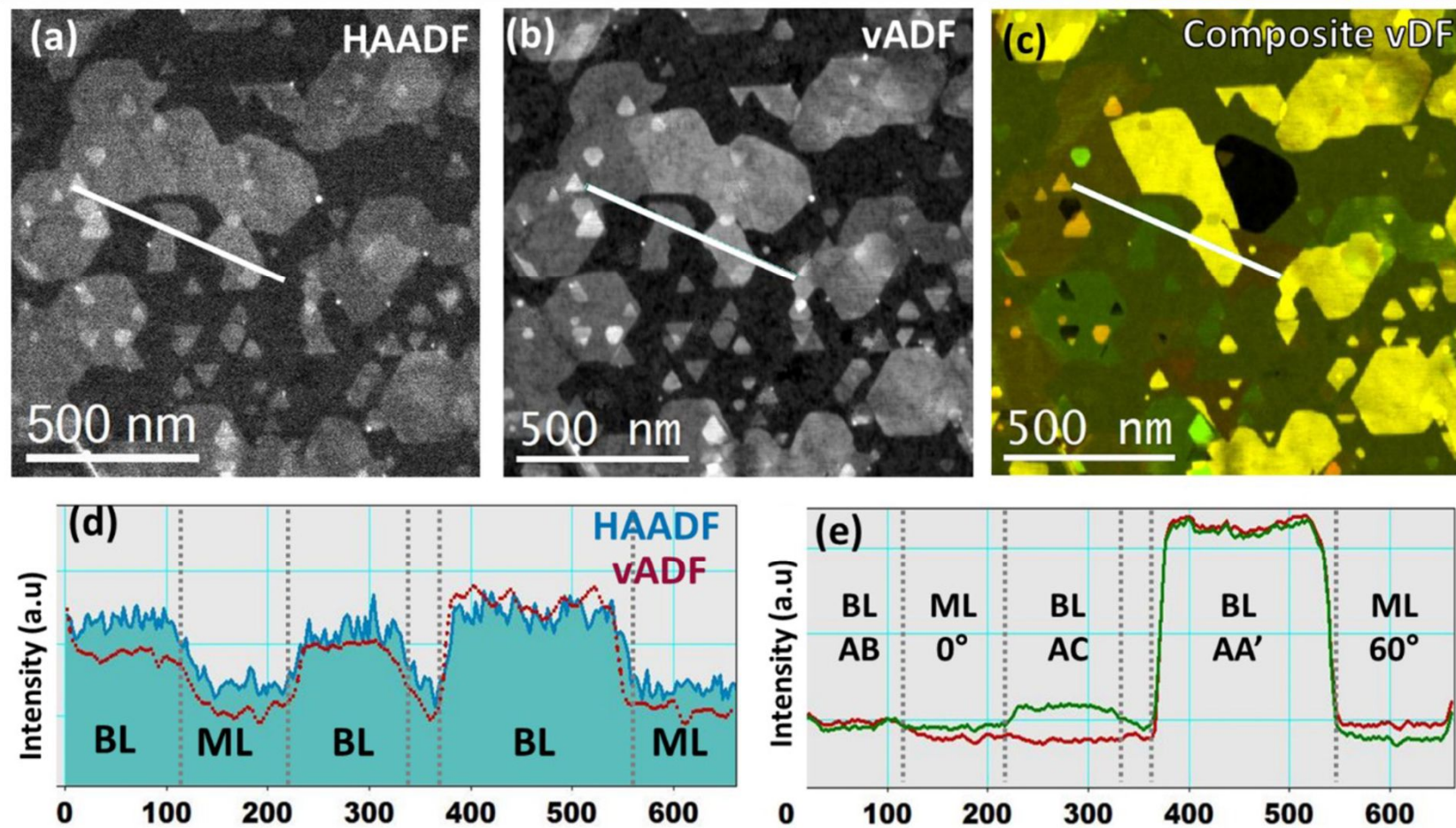


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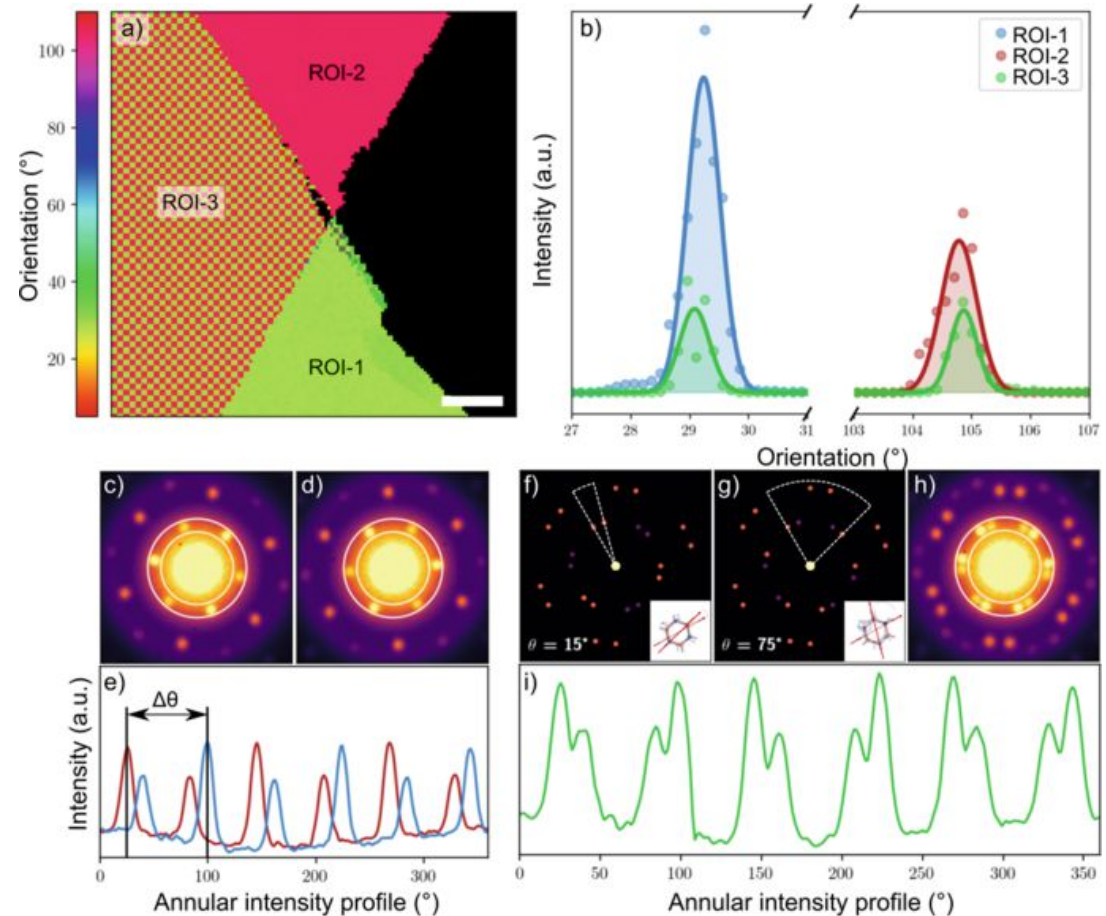
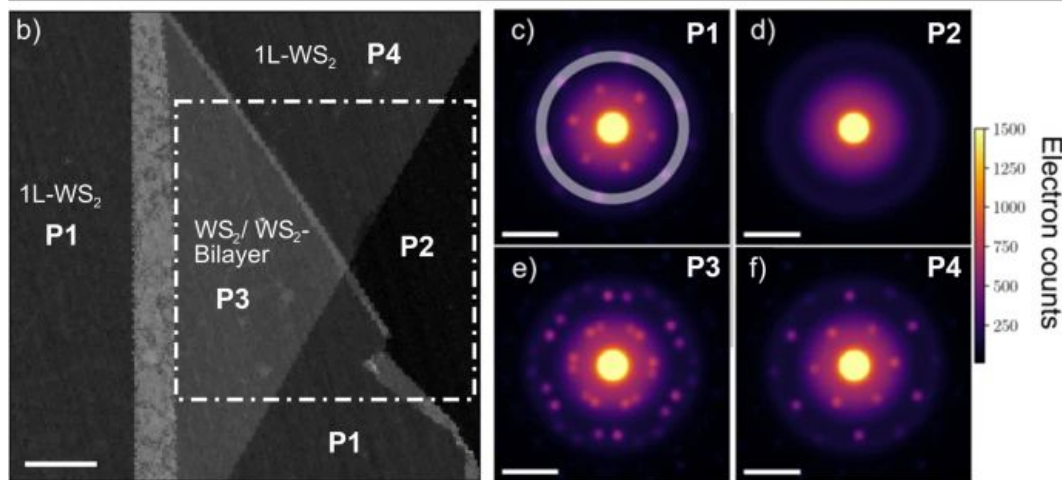
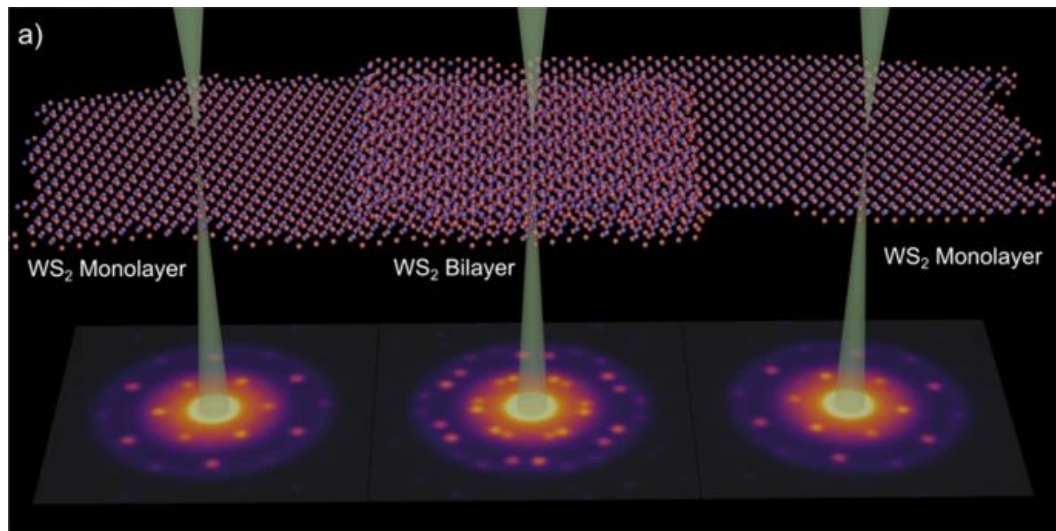
# Unraveling stacking in MoS<sub>2</sub> bilayer







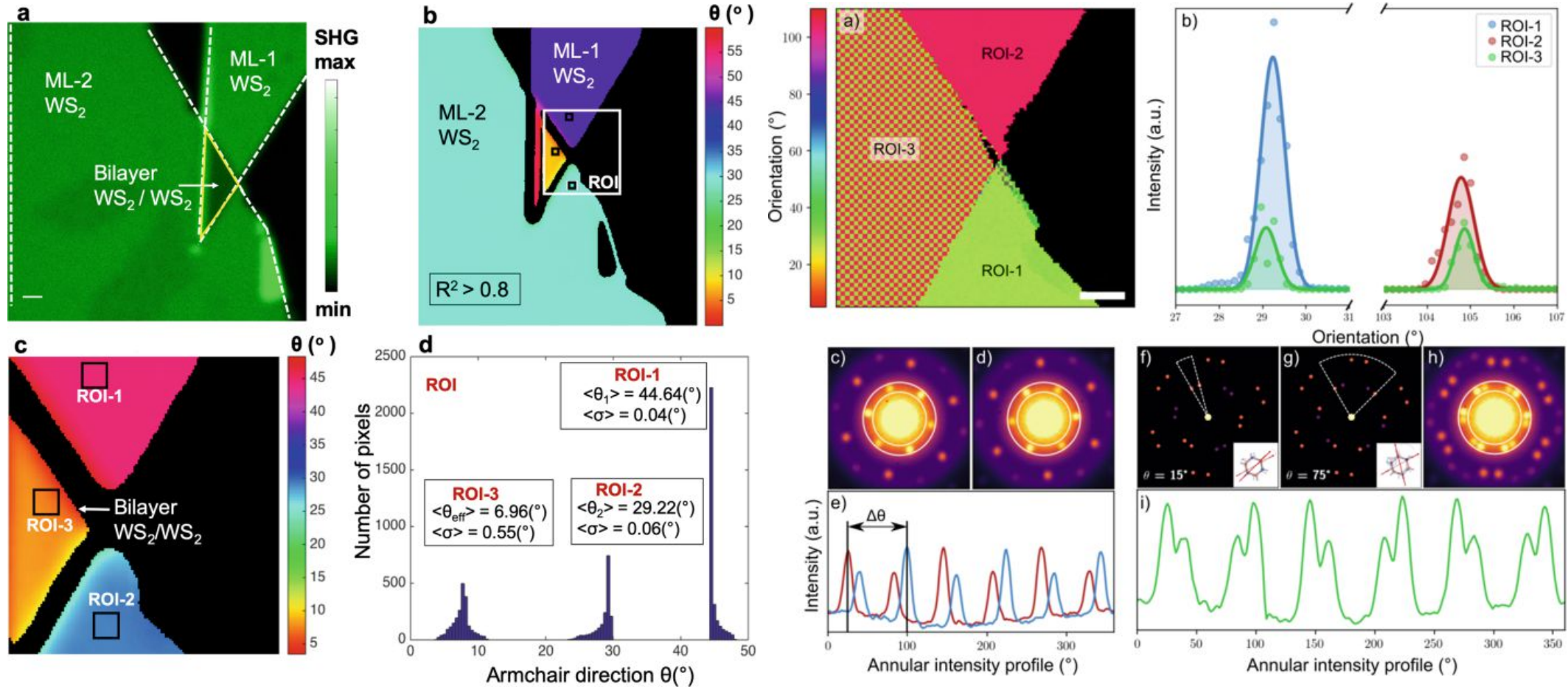
# Twist angle measurement in WS<sub>2</sub>



*S. Psilodimitrakopoulos et al., npj2D Materials and Applications, 5, 2021; 77*



# Twist angle measurement in WS<sub>2</sub>



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TEM has limited field of view (a couple of  $\mu m^2$ ) → Statistics at microscale can be rather low

- Complicated operation mode → Skilled operator with much experience
- Limited space for sample → Heating/Gas/Bias experiments in dedicated holders
- Low interaction due too high velocity electrons (for very thin materials)

→ SEM

### **Disadvantage of SEM:**

Lower spatial resolution (still better than optical )

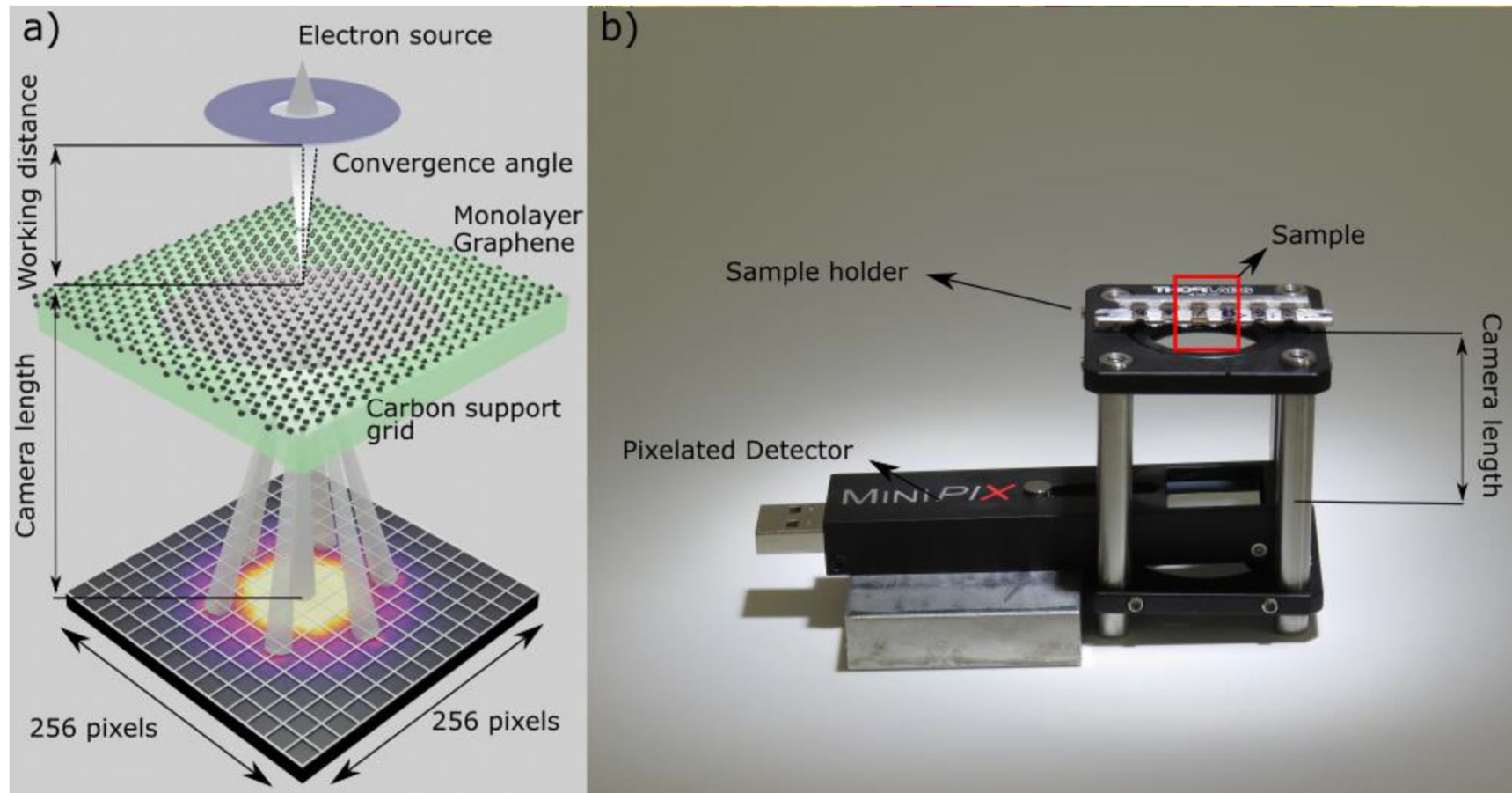
Thickness due too lower acceleration voltage

More damage at lower acceleration voltage??? → Ideal for 2D materials



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# Setup

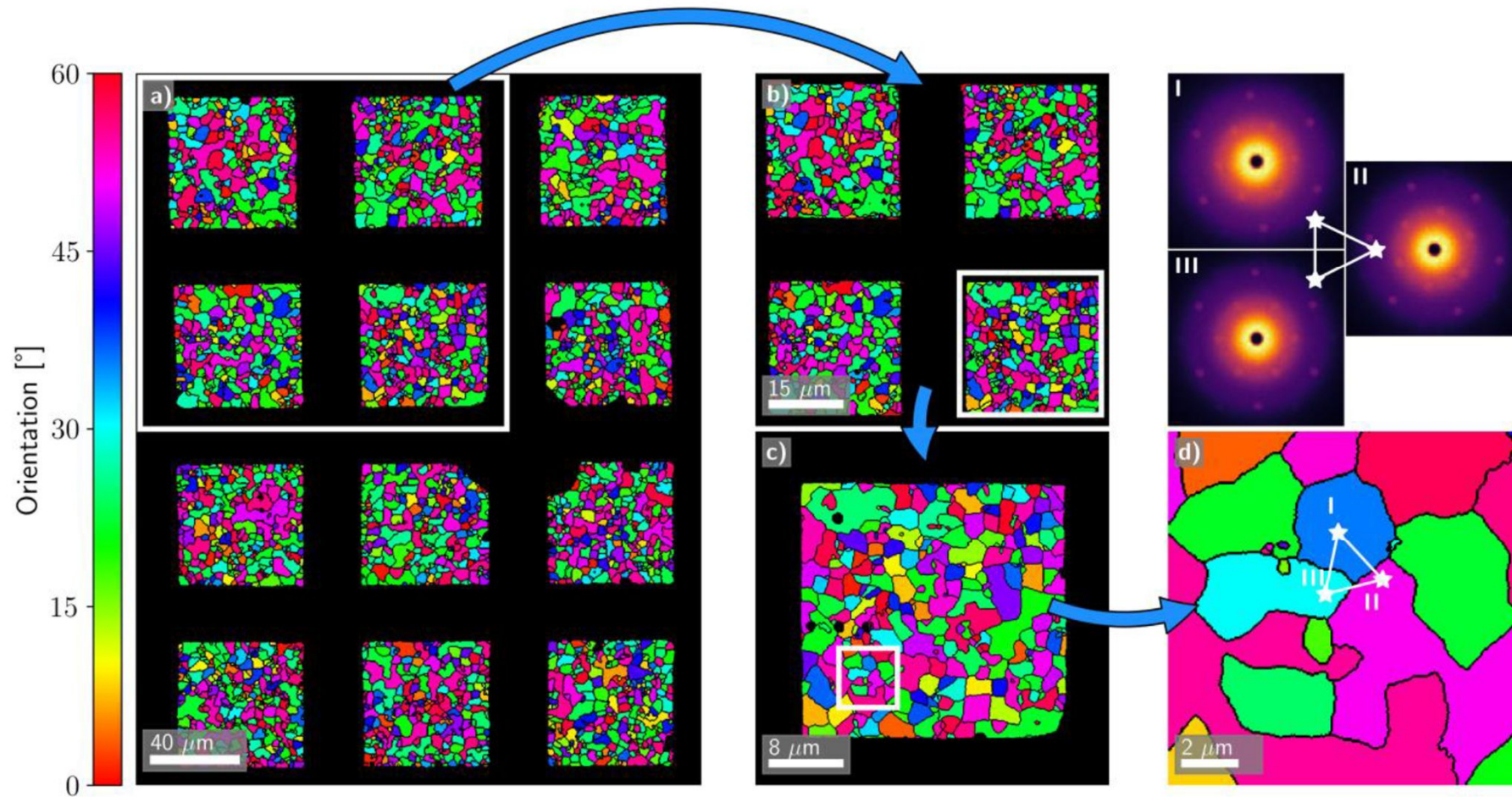


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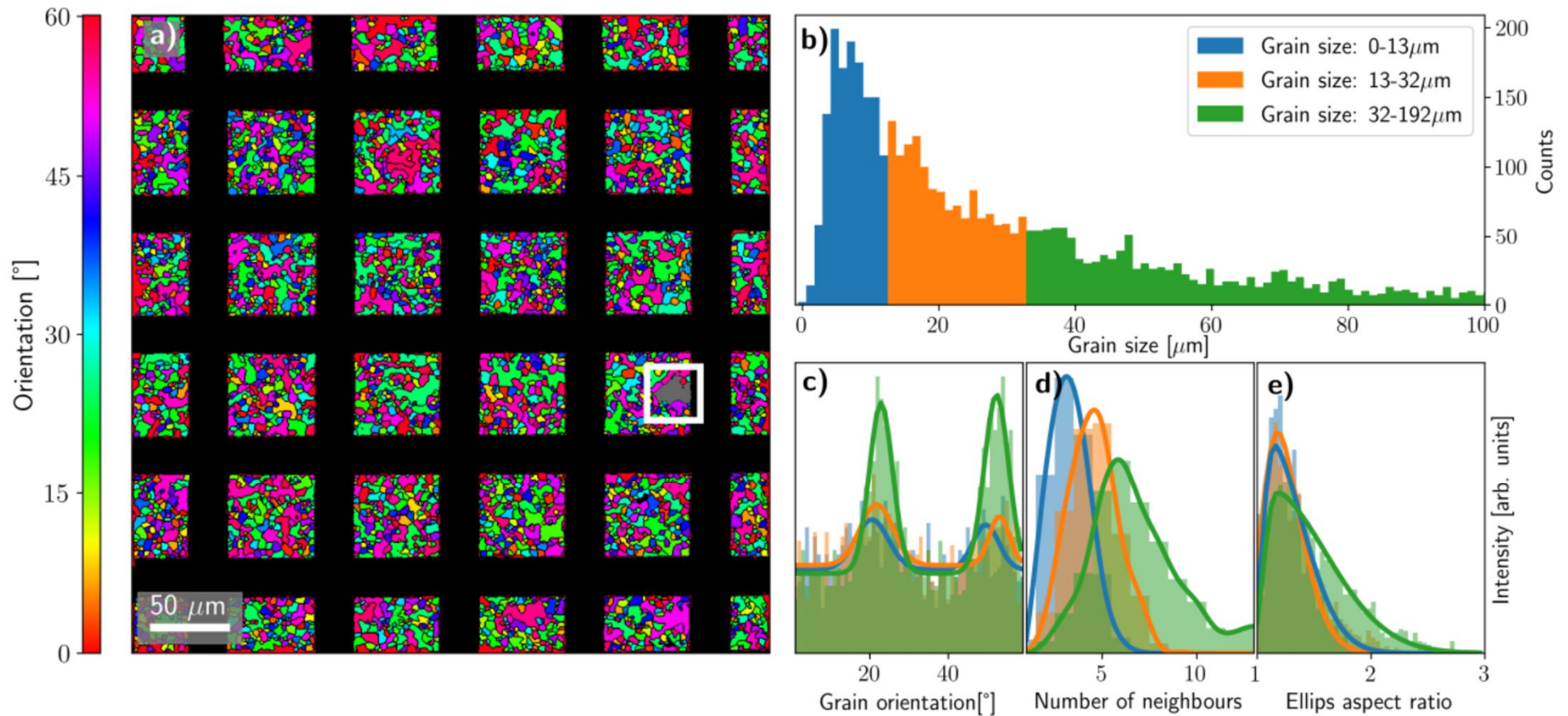
<https://doi.org/10.48550/arXiv.2011.01875>



# In the SEM large field of view



# Orientation mapping in large field of view 0.168 mm<sup>2</sup>



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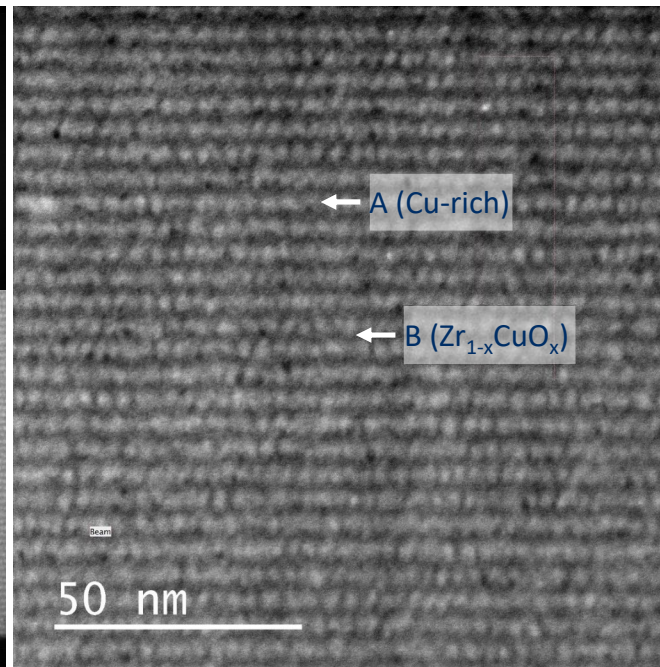
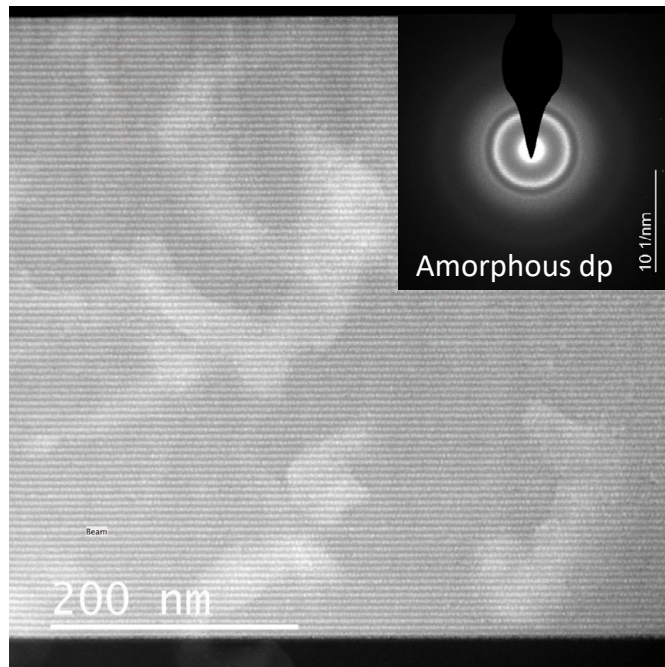
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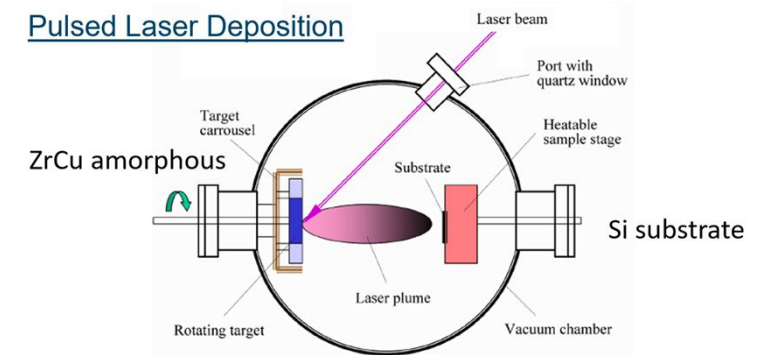
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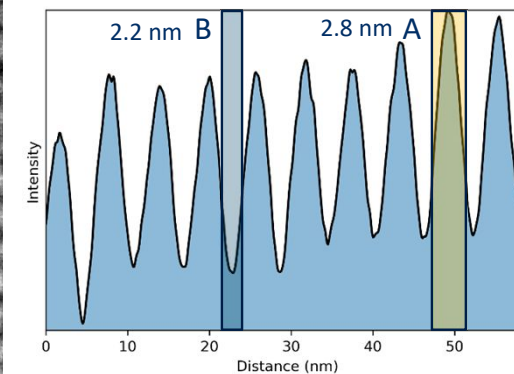
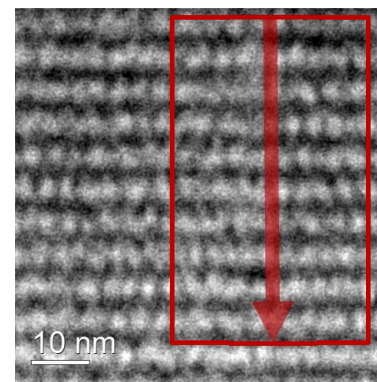
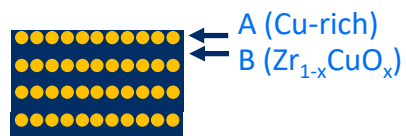
# HAADF STEM analysis of ZrCu 020 (He, P = 10 Pa)



## Pulsed Laser Deposition



## Nanostructured layered structure





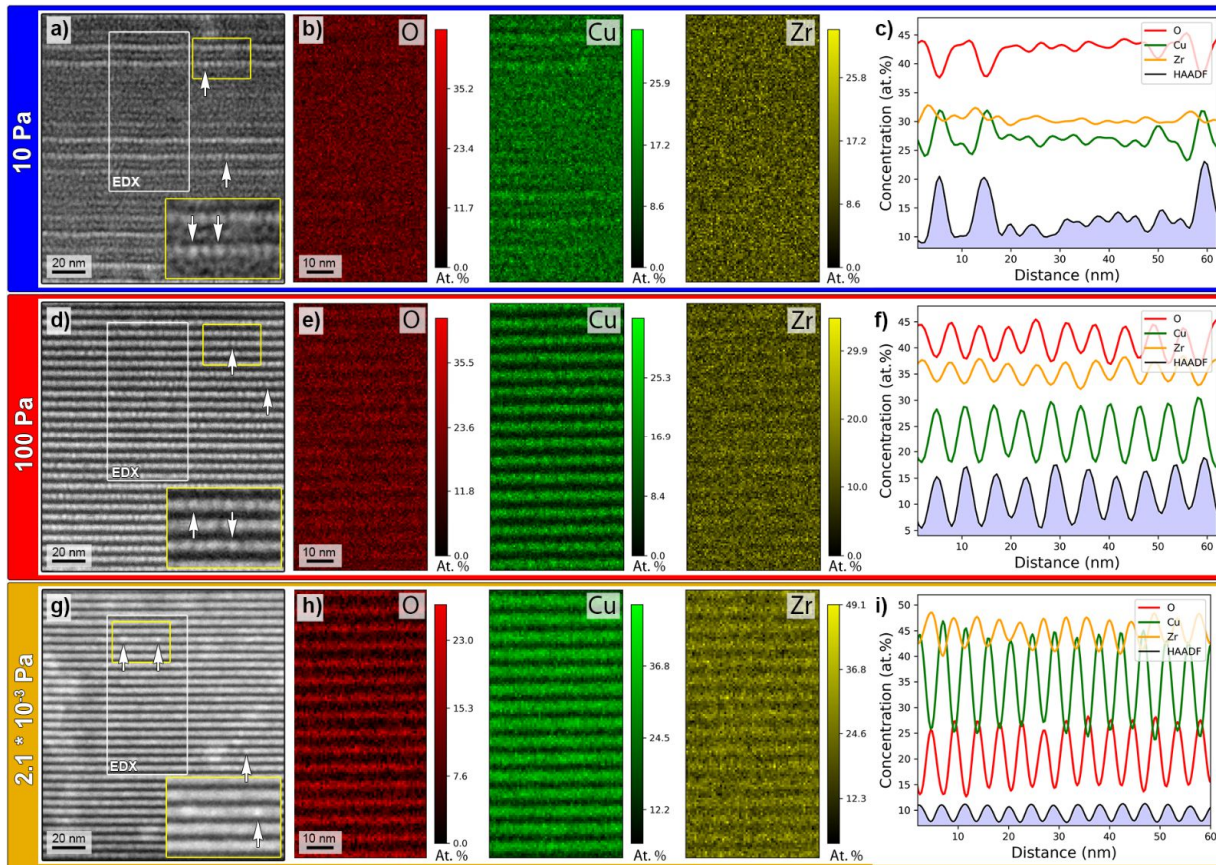
## Full length article

## Novel class of nanostructured metallic glass films with superior and tunable mechanical properties



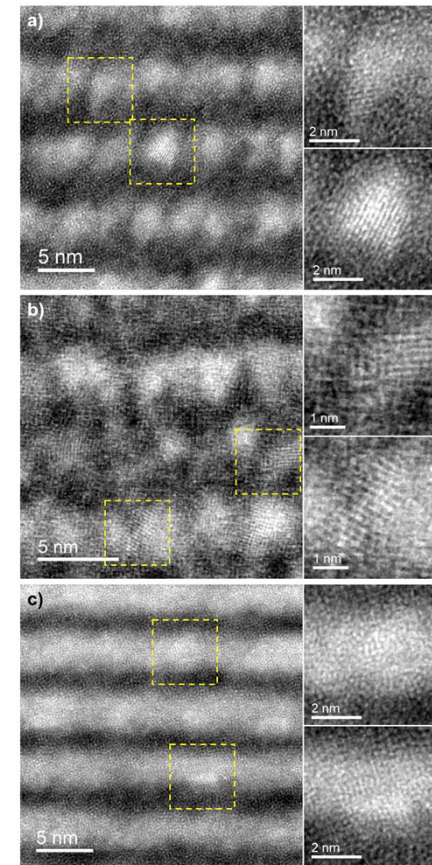
M. Ghidelli<sup>a,b,1,\*</sup>, A. Orekhov<sup>c,d,e,1</sup>, A. Li Bassi<sup>a</sup>, G. Terraneo<sup>f</sup>, P. Djemia<sup>b</sup>, G. Abadias<sup>g</sup>,  
M. Nord<sup>d,e,h</sup>, A. Béché<sup>d,e</sup>, N. Gauquelin<sup>d,e</sup>, J. Verbeeck<sup>d,e</sup>, J.-P. Raskin<sup>i</sup>, D. Schryvers<sup>d,e</sup>,  
T. Pardoen<sup>c</sup>, H. Idrissi<sup>c,d,\*</sup>

## EDX



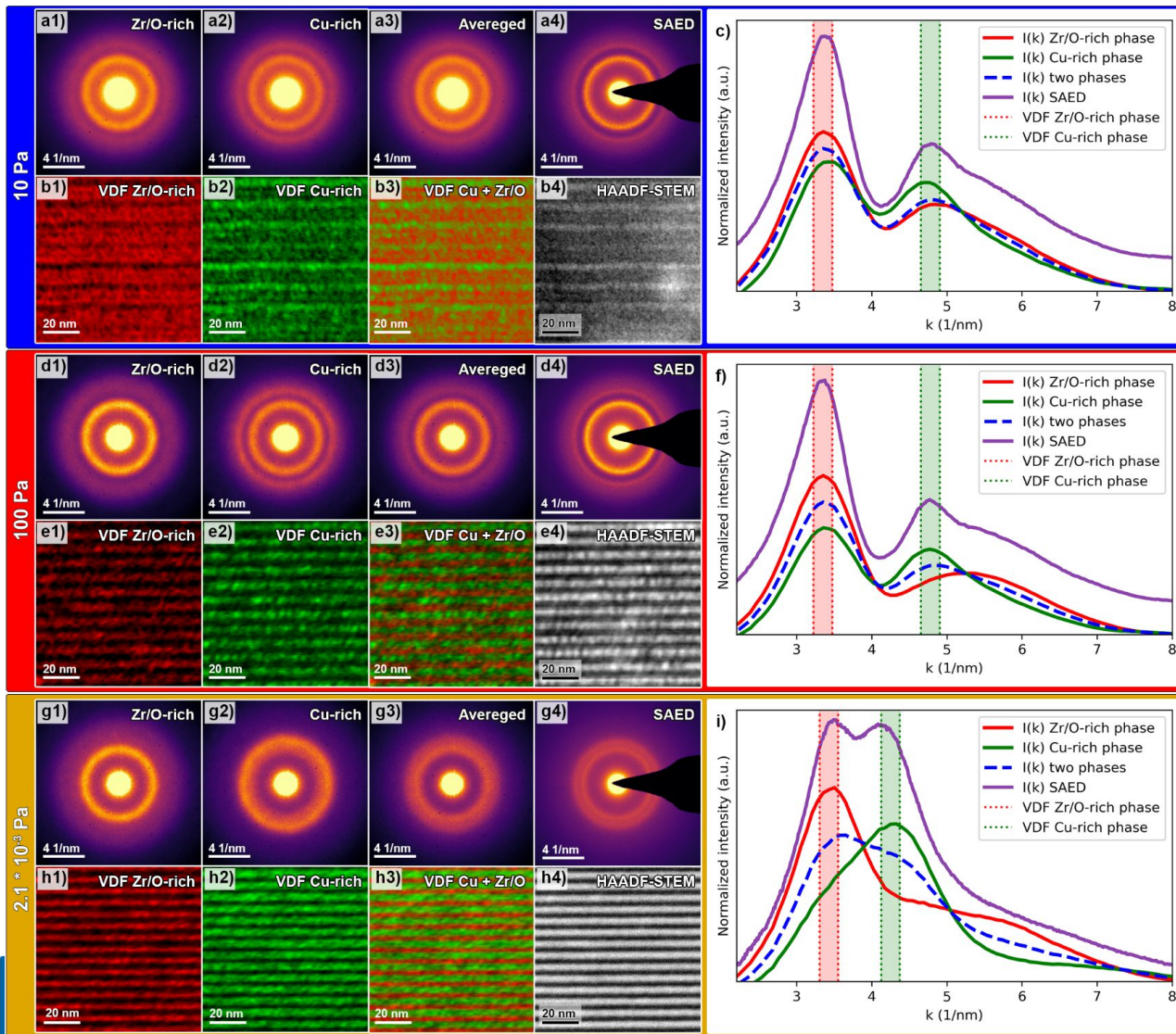
M. Ghidelli, A. Orekhov, A.L. Bassi, G. Terraneo, P. Djemia, G. Abadias, M. Nord, A. Beche, N. Gauquelin, J. Verbeeck, J.P. Raskin, D. Schryvers, T. Pardoen, H. Idrissi, Novel class of nanostructured metallic glass films with superior and tunable mechanical properties, *Acta Mater.* 213 (2021), 116955.

## HRSTEM

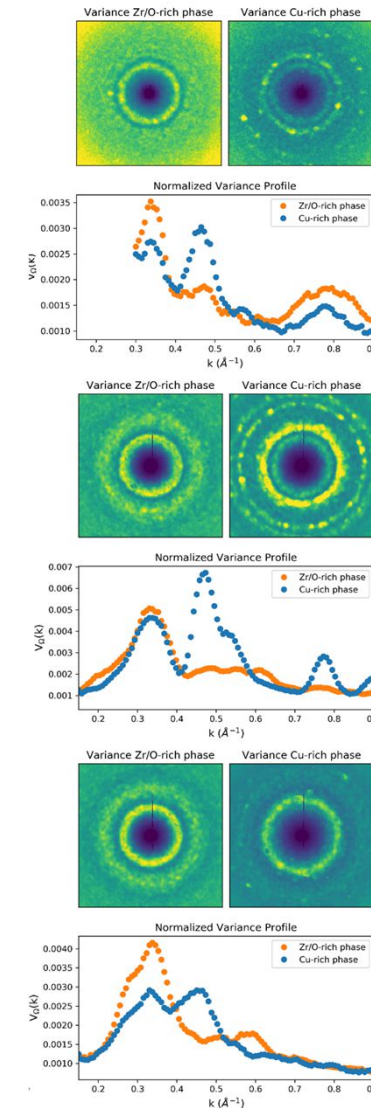




## 4D STEM data



## Variance diffraction



crystalline clusters  
(1-2 nm) in the Cu-  
rich phase

Zr/O-rich phase  
exhibits a more  
'disordered'  
structure

Cu-rich phase single peak  
shifted to higher  $k \rightarrow$   
changes of the nature of  
local order (no clusters)

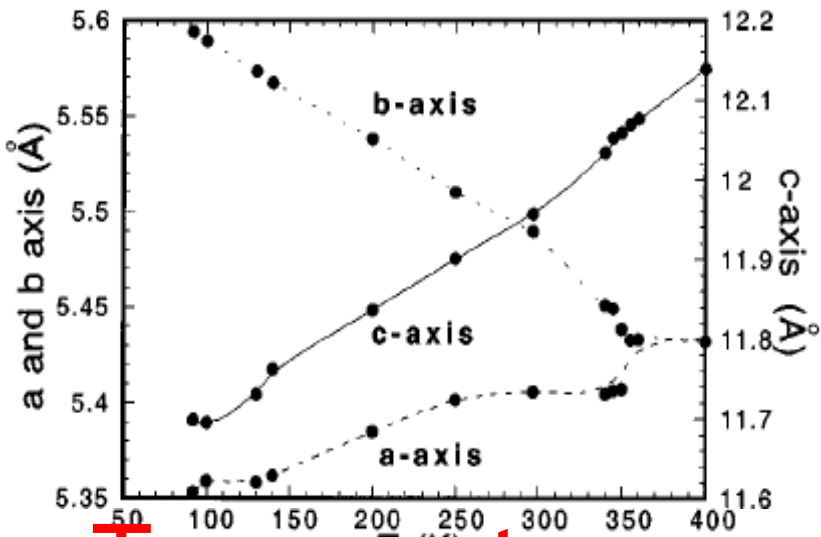
Zr/O-rich phase similar  
"disordered"

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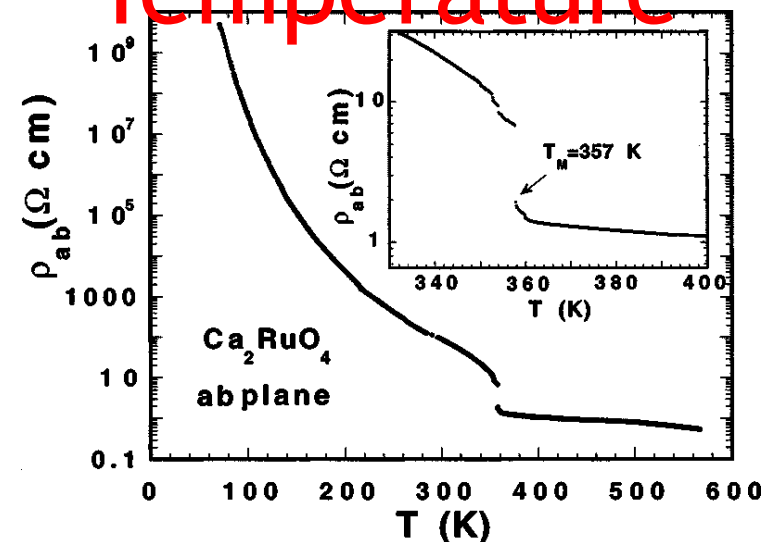
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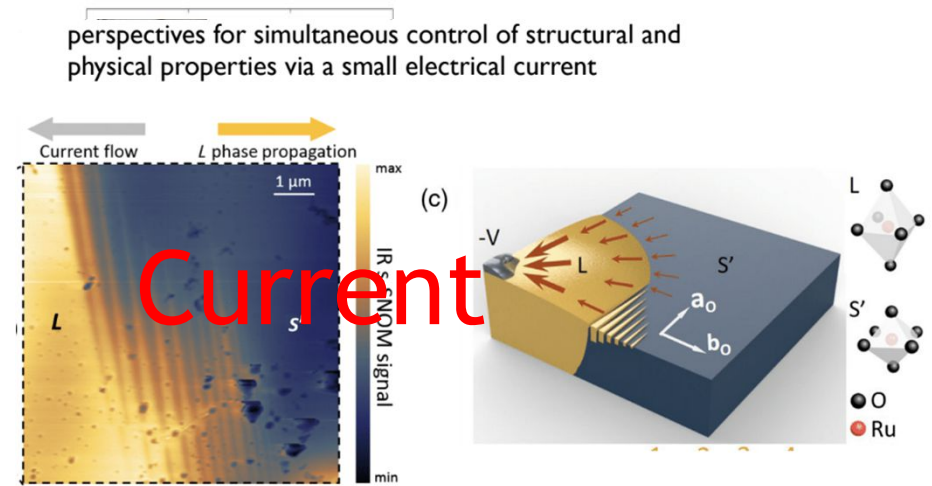
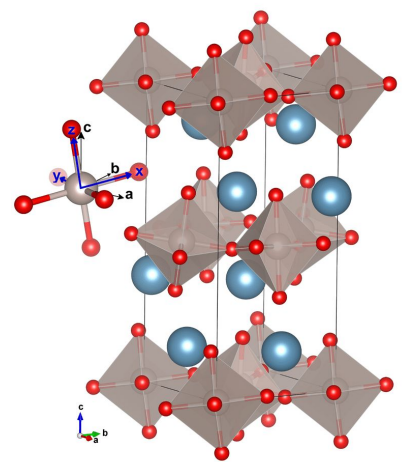
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Temperature

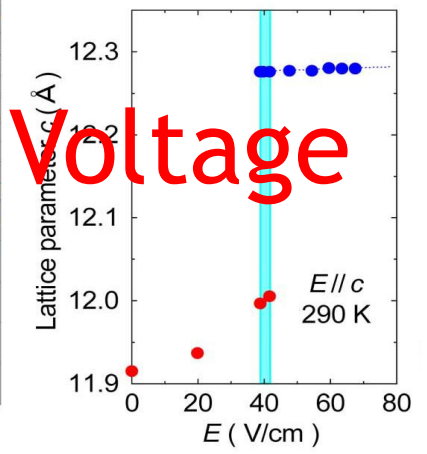
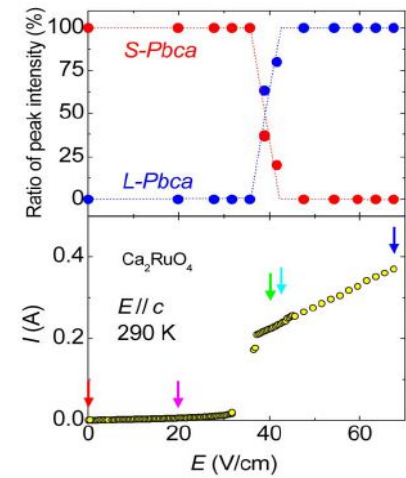
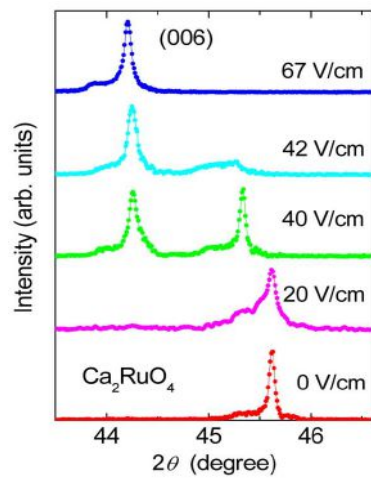


Phys. Rev. B 60, R8422 (1999)



Phys. Rev. X 9, 011032 (2019)

Insulator to metal phase transition happening with T, I or V

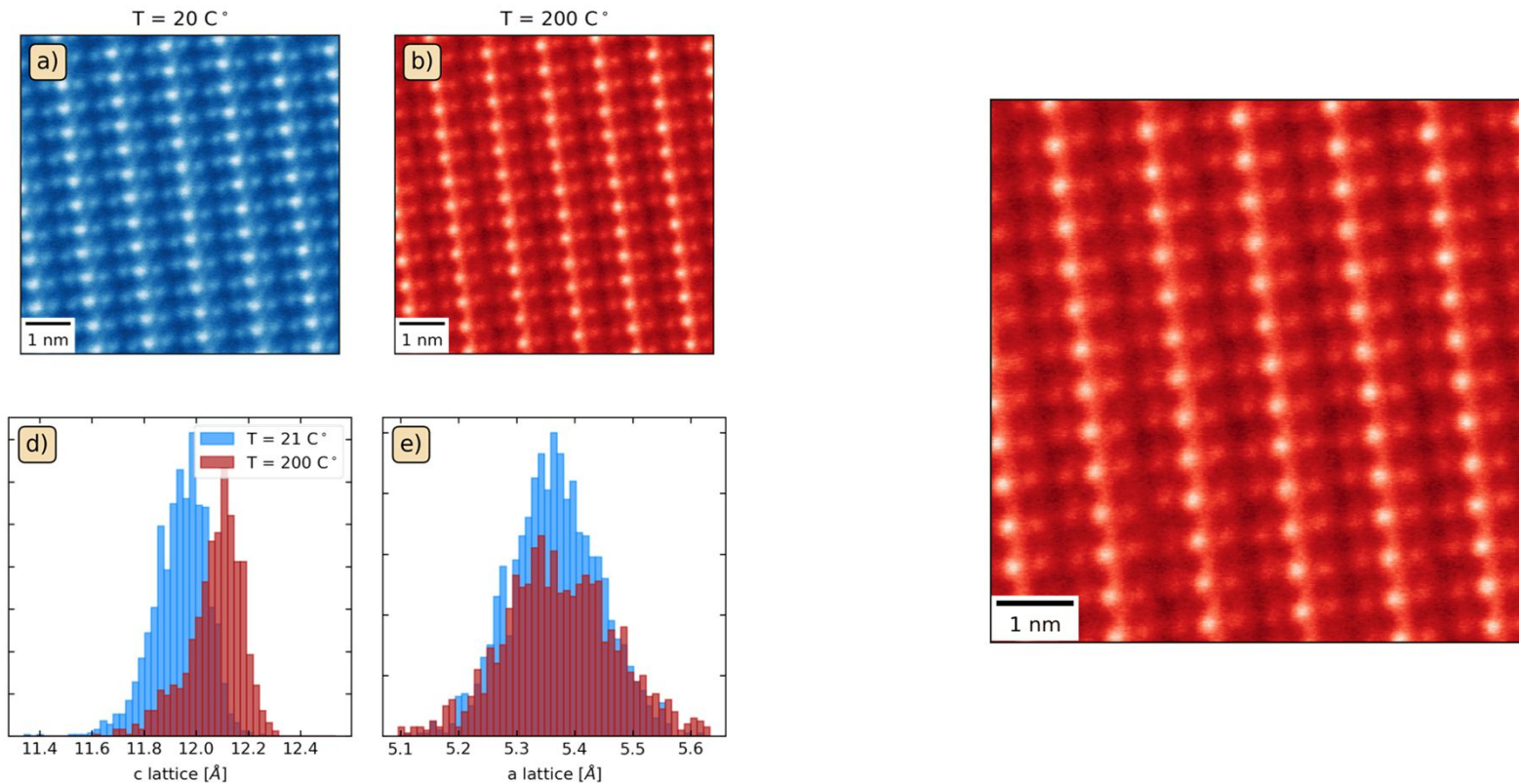


Voltage

F. Nakamura et al., Sci. Rep. 3, 2536 (2013)



# Temperature induced phase transition

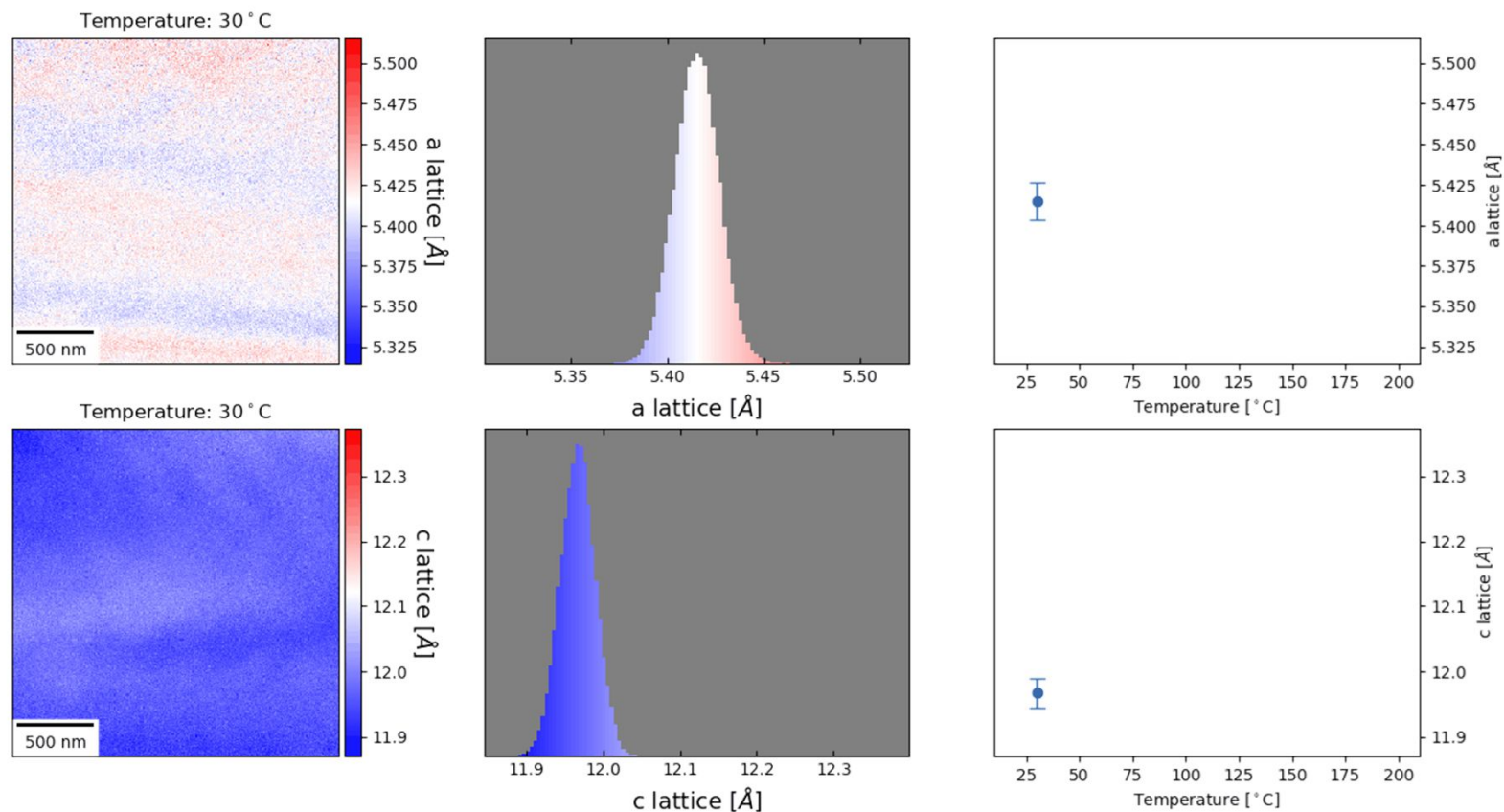


N. Gauquelin, F.Forte et al., Nano Letters (2023) <https://doi.org/10.1021/acs.nanolett.3c00574>

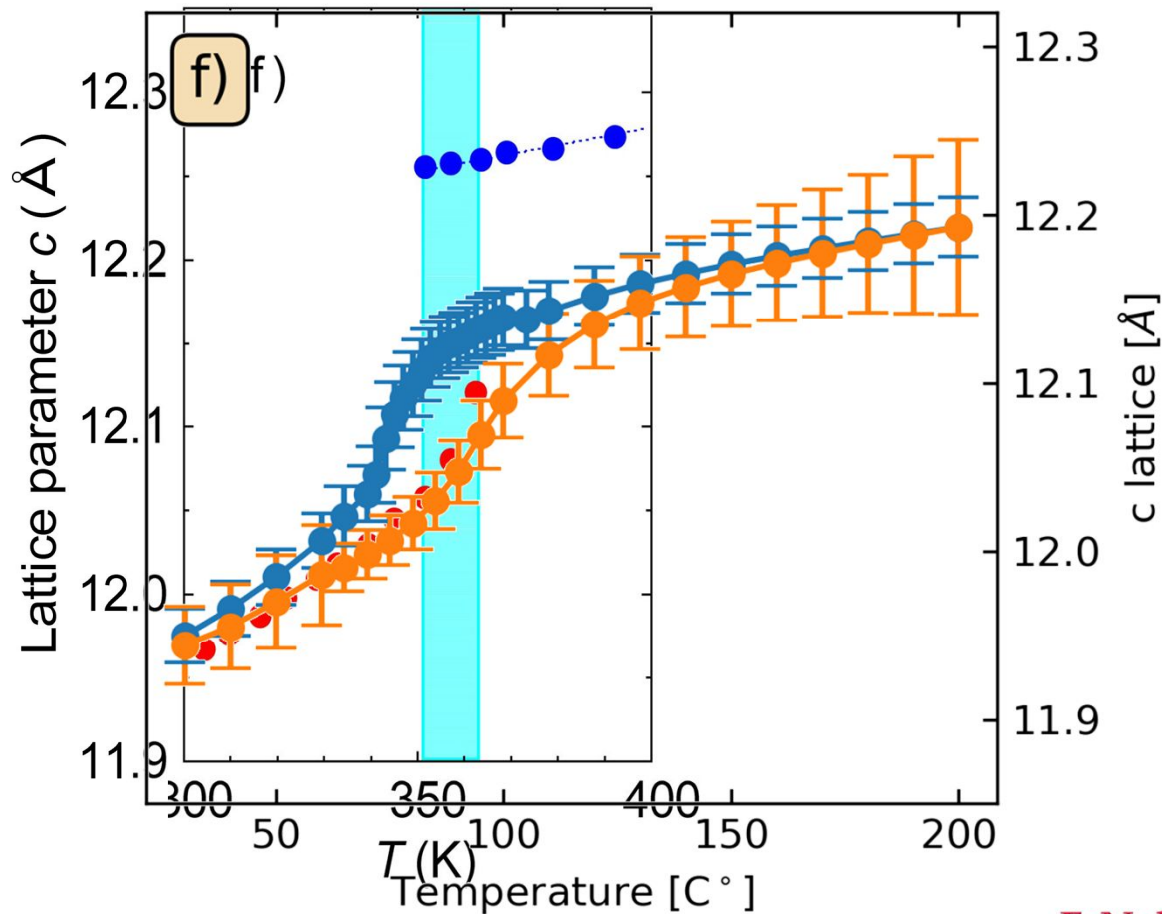


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# Experiment: Temperature experiment

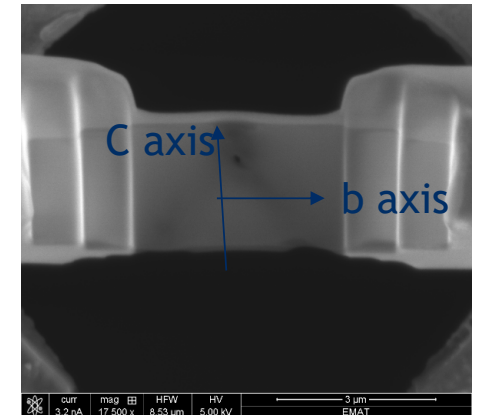


# Experiment: Temperature experiment



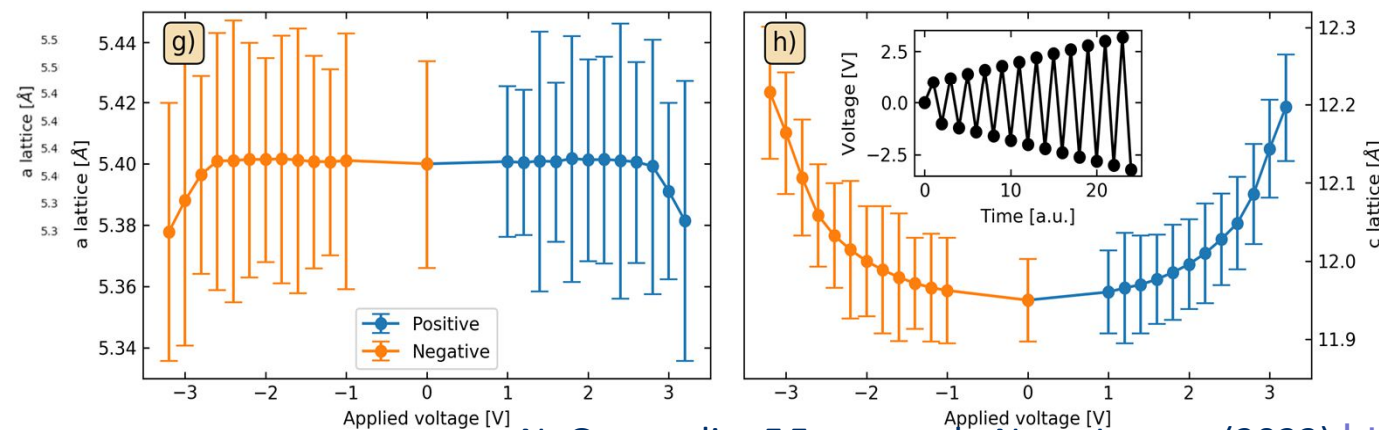
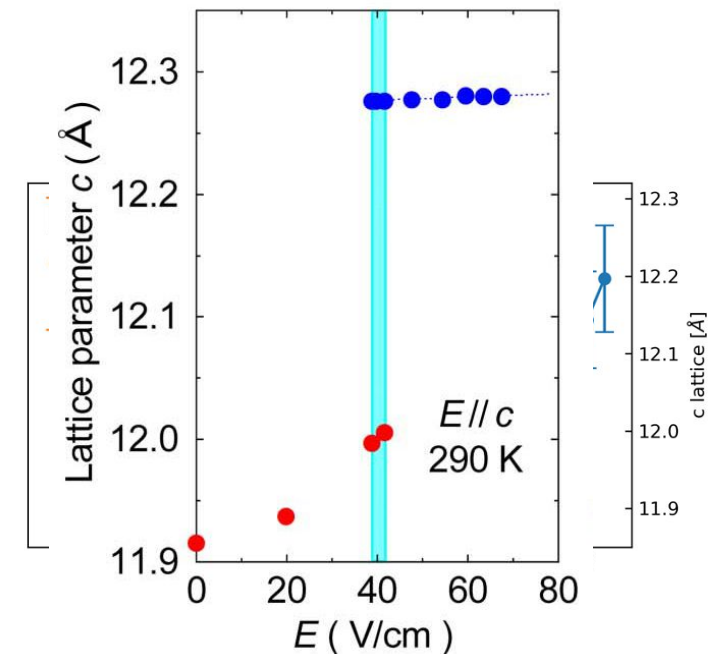
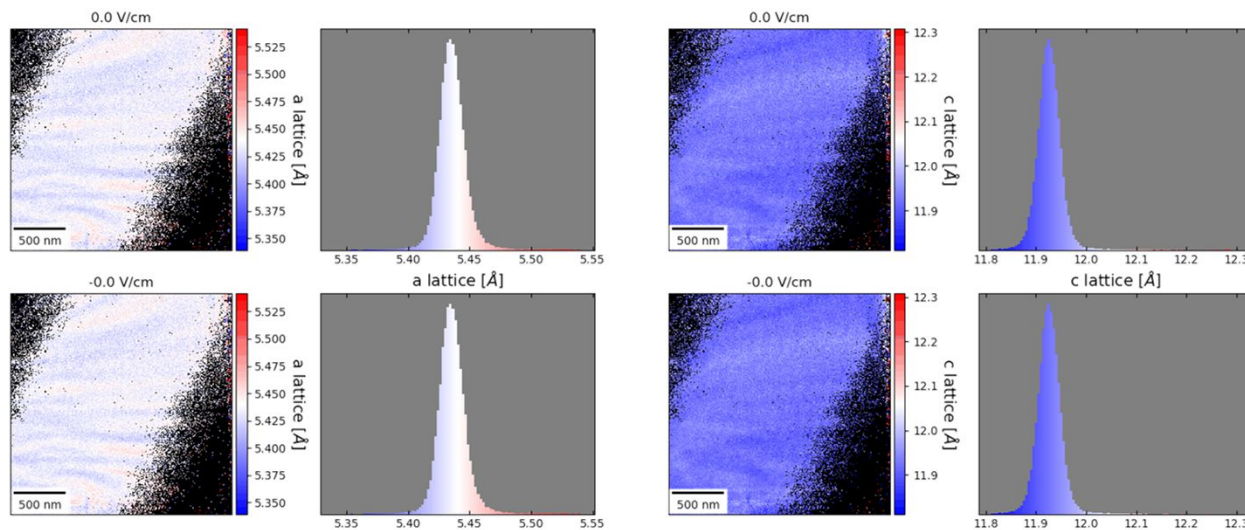
We do not reach the same value of  $c$ -lattice parameter than Nakamura et al.

Strain induced by the geometry



F. Nakamura et al., Sci. Rep. 3, 2536 (2013)

# Biasing experiment



Voltage sequence:  
 $0 \rightarrow 1000\text{mV} \rightarrow -1000\text{mV} \rightarrow 1200\text{mV}$   
 $\rightarrow \dots \rightarrow -3200\text{mV}$

Conclusion: No stripe pattern is observed using this voltage sequence

N. Gauquelin, F.Forte et al., Nano Letters (2023) <https://doi.org/10.1021/acs.nanolett.3c00574>

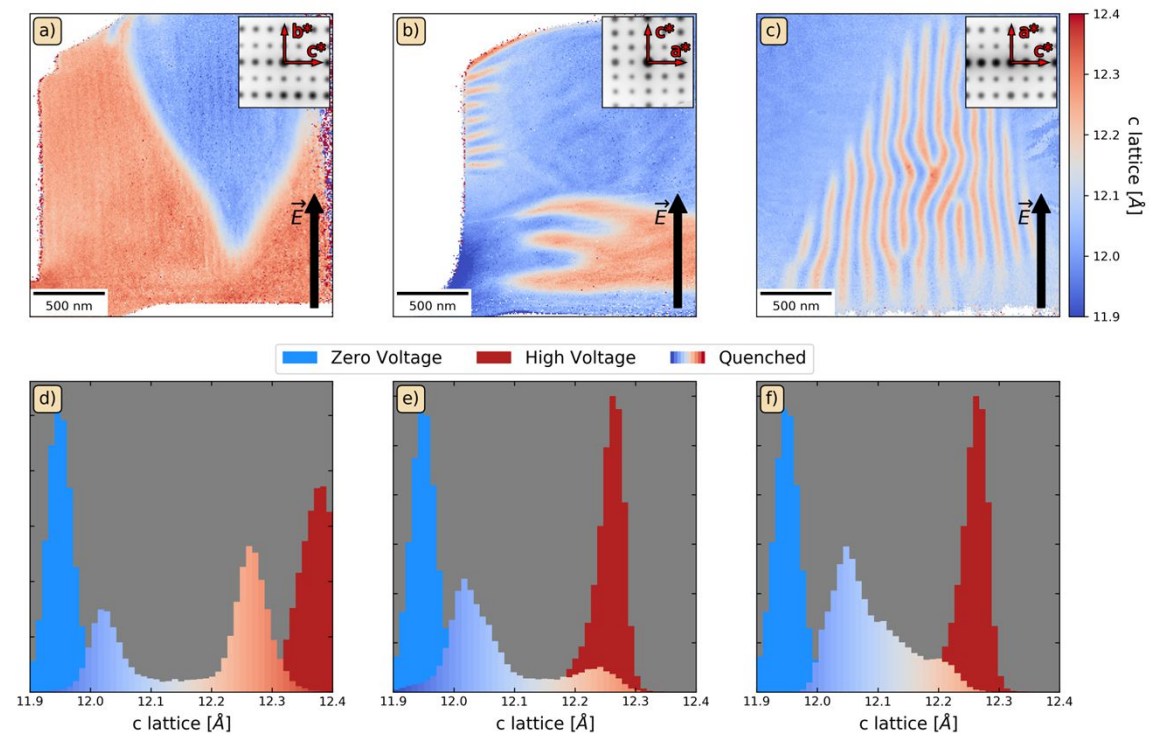


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# After electric field quench different orientations

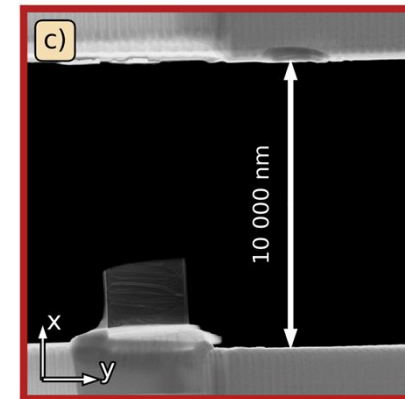
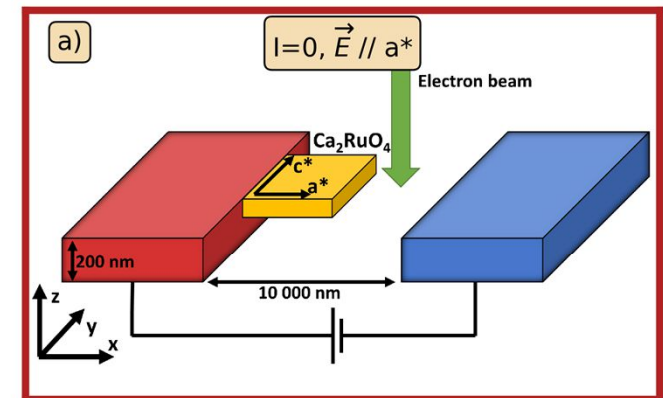
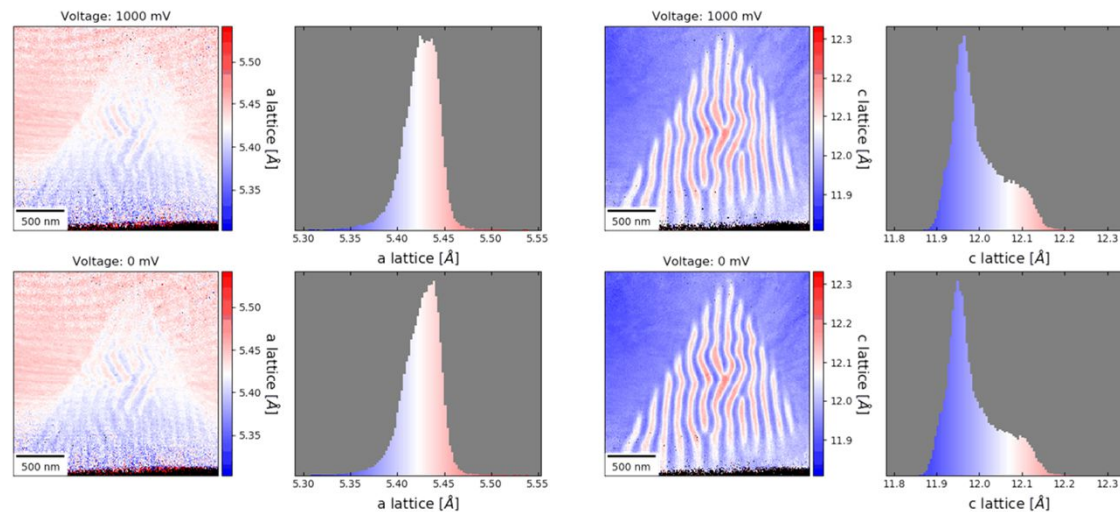
- After quenching electric field, different patterns arise when changing crystal orientation with respect to electric field



N. Gauquelin, F.Forte et al., Nano Letters (2023) <https://doi.org/10.1021/acs.nanolett.3c00574>

# Evolution lattice parameters

- 4D STEM when applying voltage and quenching after
  1. row: By applying voltage domains can be modified
  2. row: Quench after brings back to stripe phase

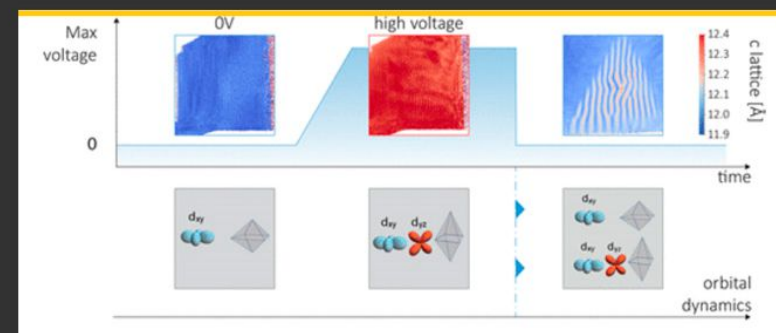




# NANO LETTERS

## Pattern Formation by Electric-Field Quench in a Mott Crystal

**Nicolas Gauquelin\***, Filomena Forte, **Daen Jannis**, Rosalba Fittipaldi, Carmine Autieri, Giuseppe Cuono, Veronica Granata, Mariateresa Lettieri, Canio Noce, Fabio Miletto-Granozio, Antonio Vecchione\*, **Johan Verbeeck**, and Mario Cuoco



**Article | Published: 13 September 2023**

# Outline

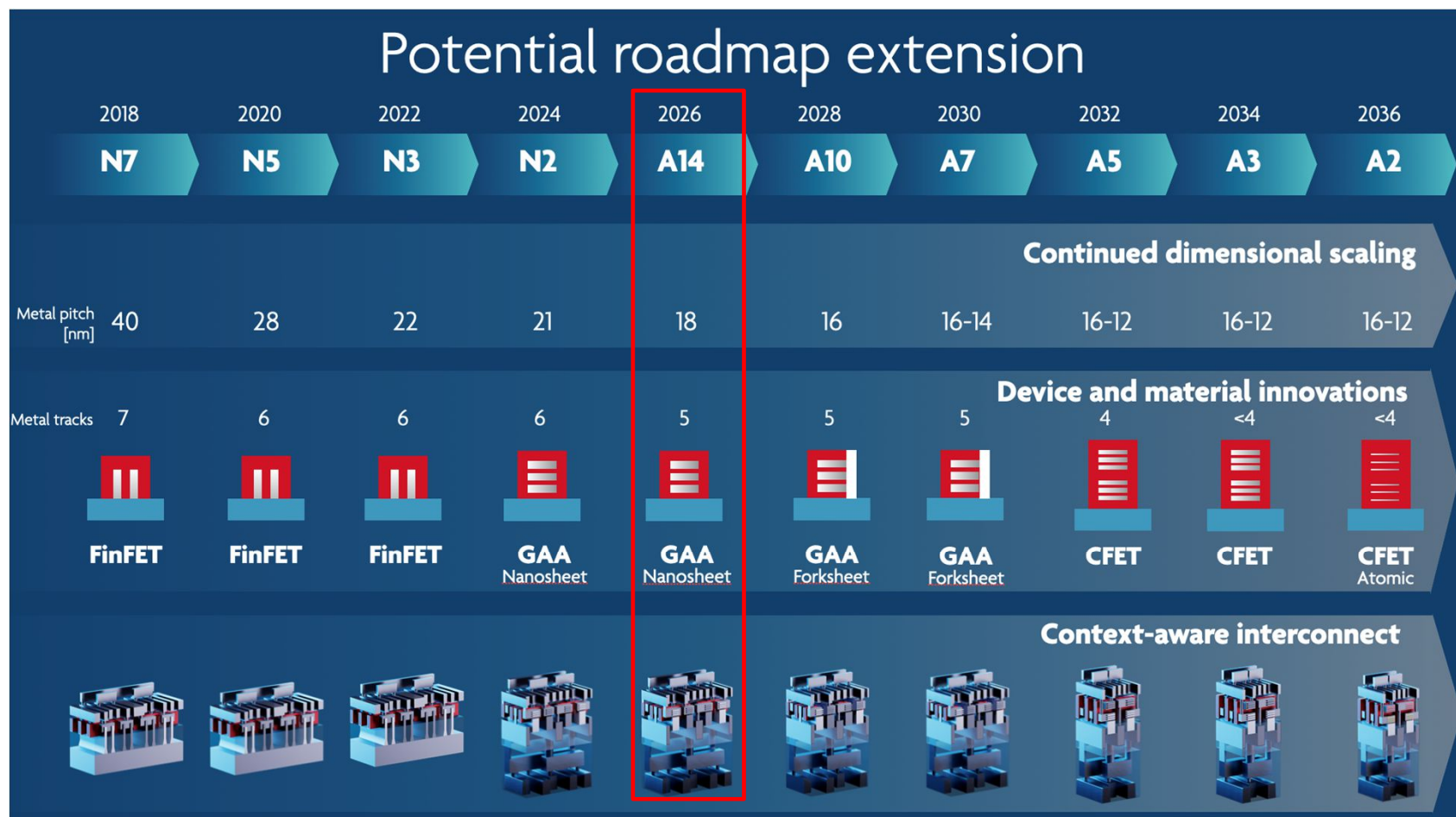
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  - EELS-EDX tomography
  - 4DSTEM
- Perspectives and outlook



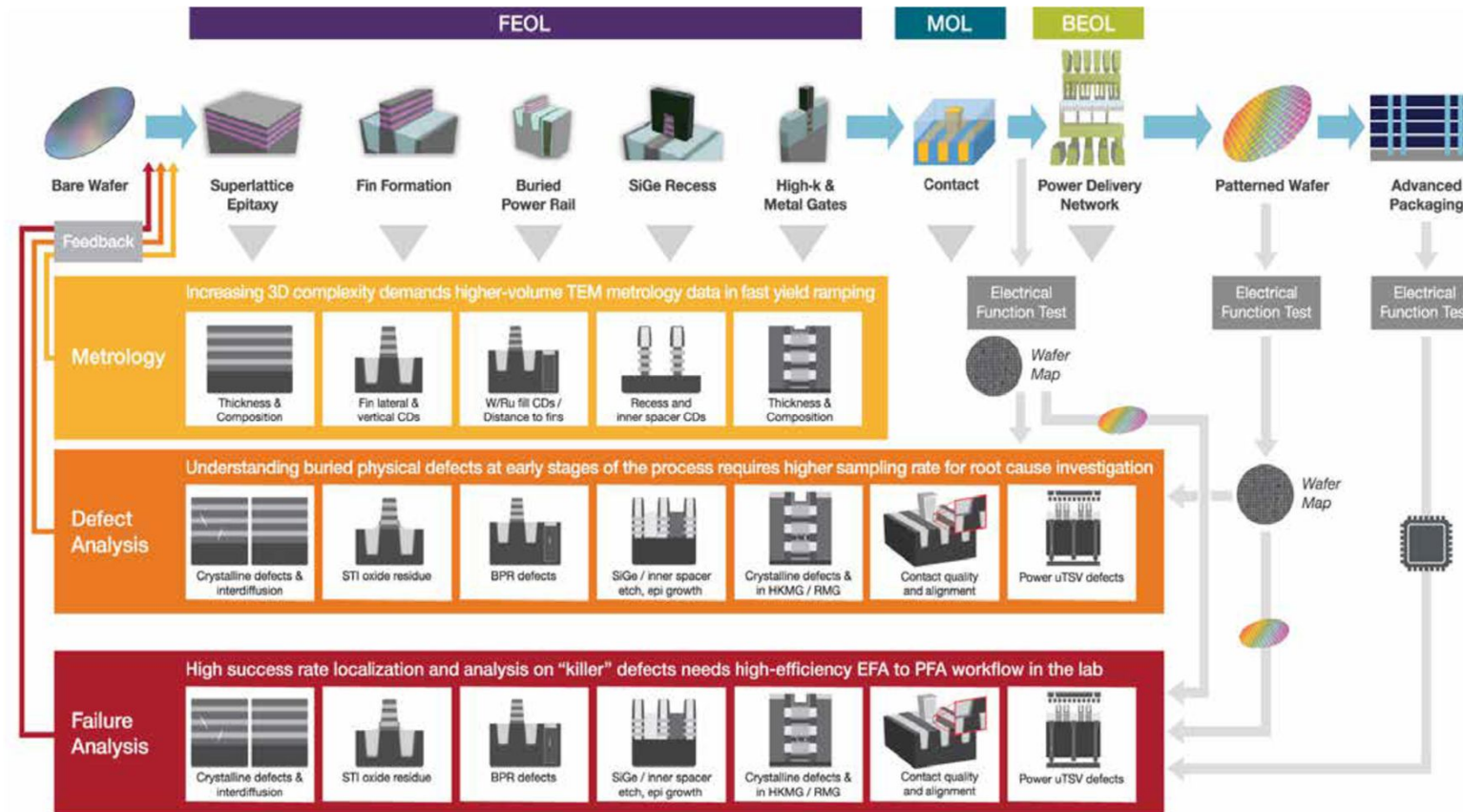
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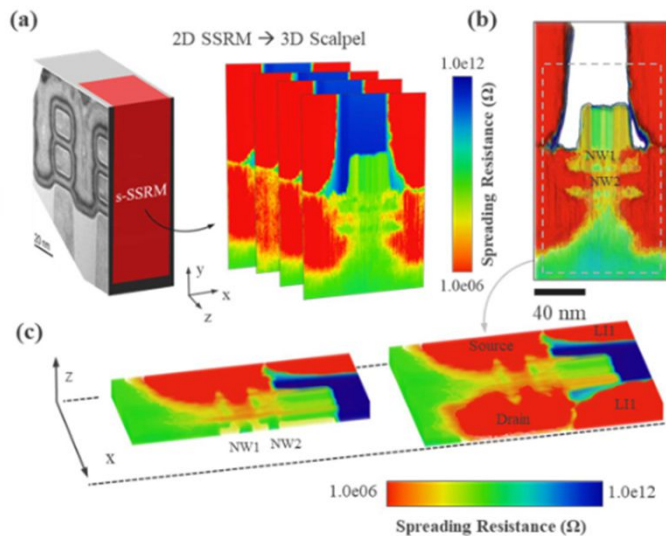
# Increasing complexity of semiconductor technology



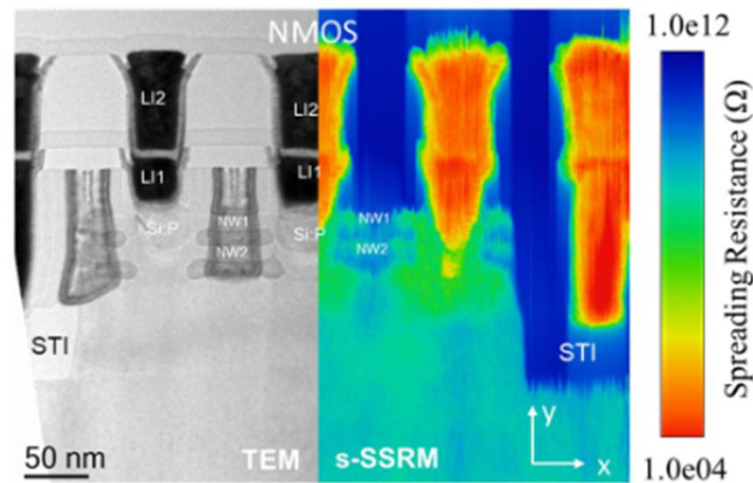
# Challenges in metrology of GAA-FET



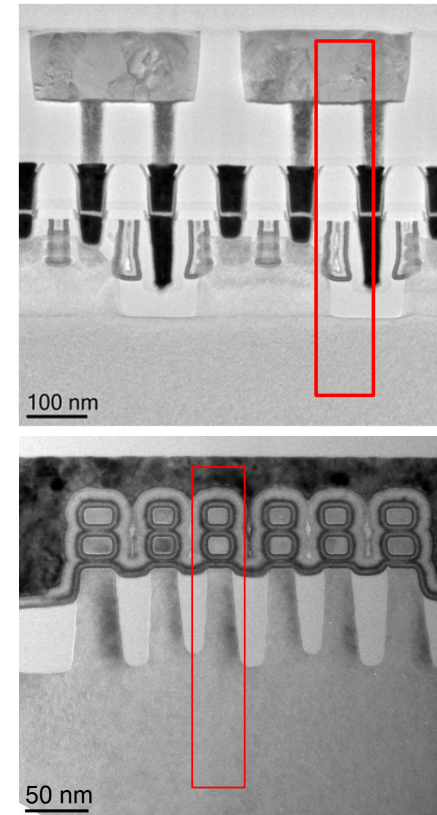
# pMOS GAA Si NW-FET sample



**Fig. 5.** 3D resistance map obtained by scalpel SSRM technique stacking SSRM 2D images for the pMOS GAA Si NW-FET. Highly conductive volumes are set in red, non-conductive in blue. Local interconnect, source and drain, as well as nanowires surrounded by the metal gates can be identified. Carrier distribution into NWs can be analyzed.



**Fig. 6.** (left) TEM cross-section along the NWs of the nMOS GAA Si NW-FET. (right) Scalpel SSRM 2D spreading resistance map at low contact force for the same device showing the diffusion of dopants into the NWs. Metals are set in red, highly conductive Si is in light green.



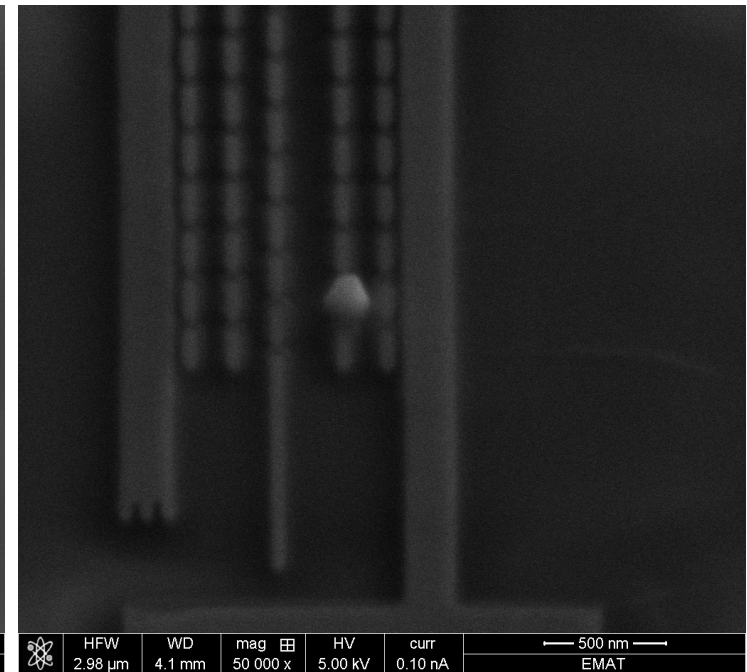
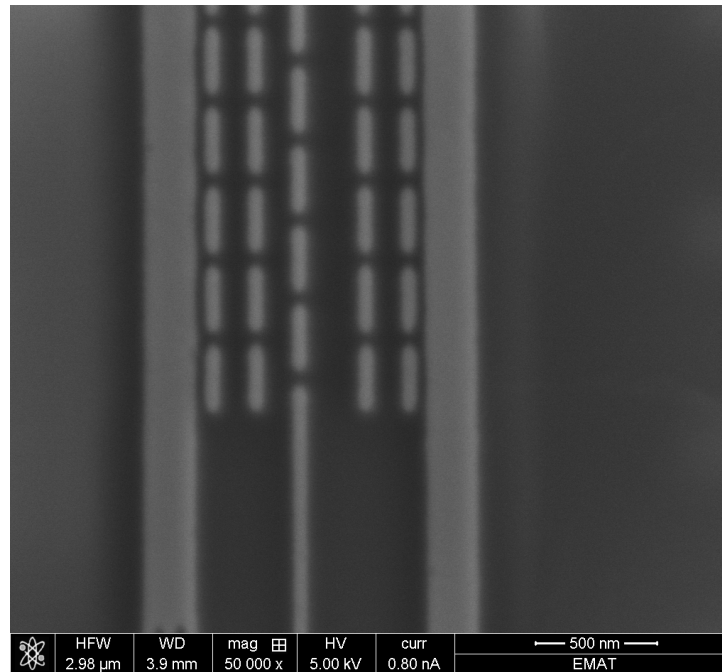
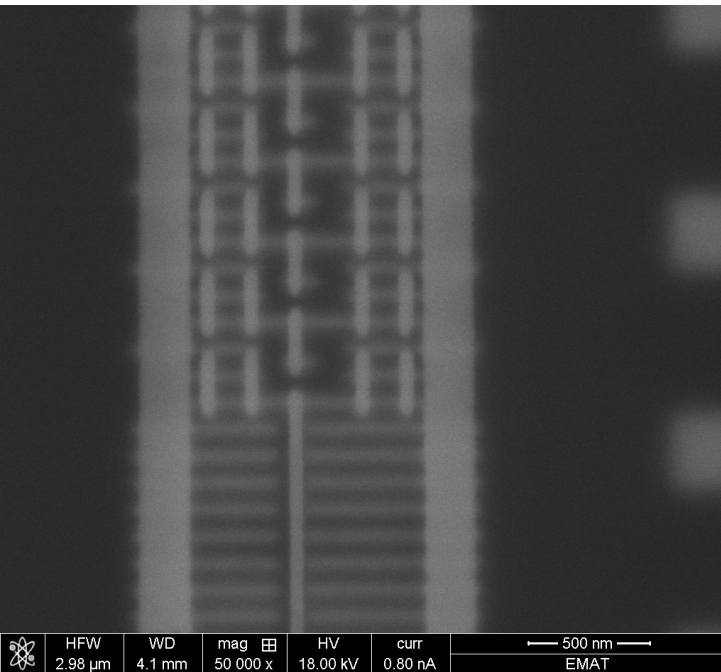
**3D-carrier Profiling and Parasitic Resistance Analysis in Vertically Stacked Gate-All-Around Si Nanowire CMOS Transistors -**  
Conference: 2019 IEEE International Electron Devices Meeting (IEDM)



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# Preparation of the needle

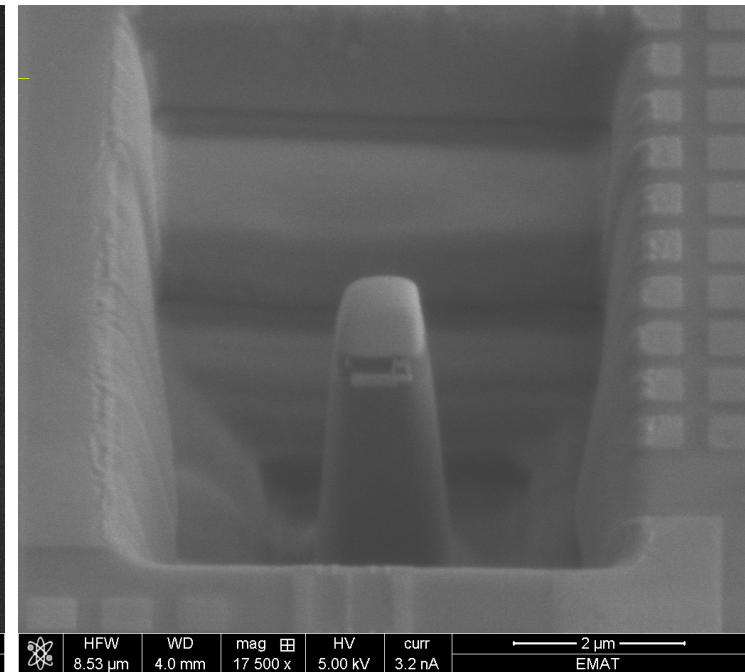
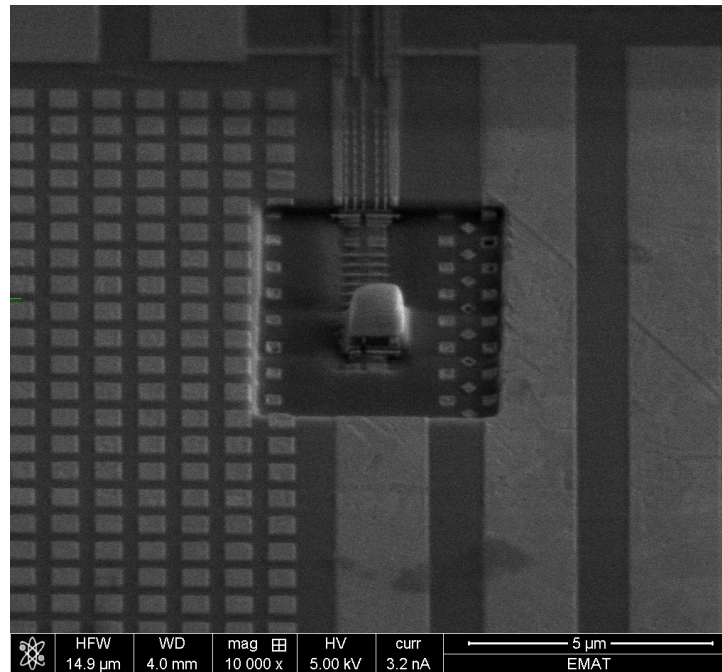
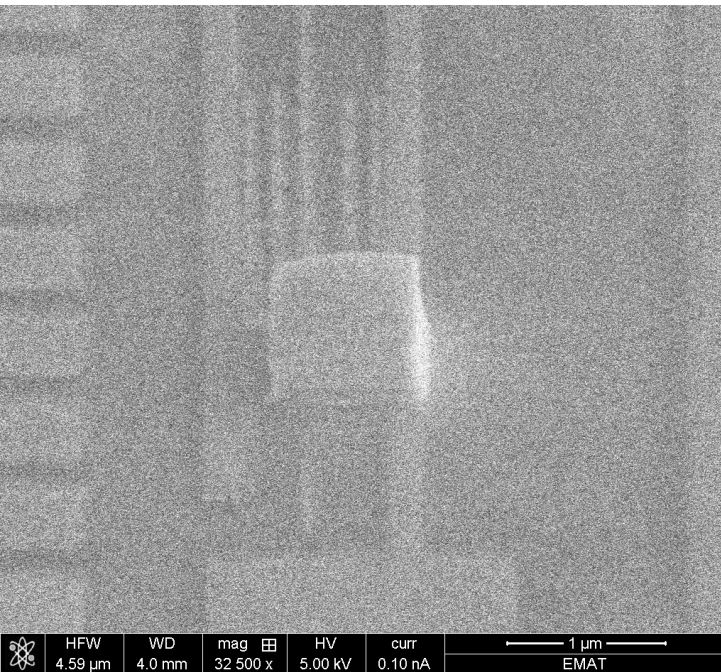


Imaging with different voltages → sensitivity to depth

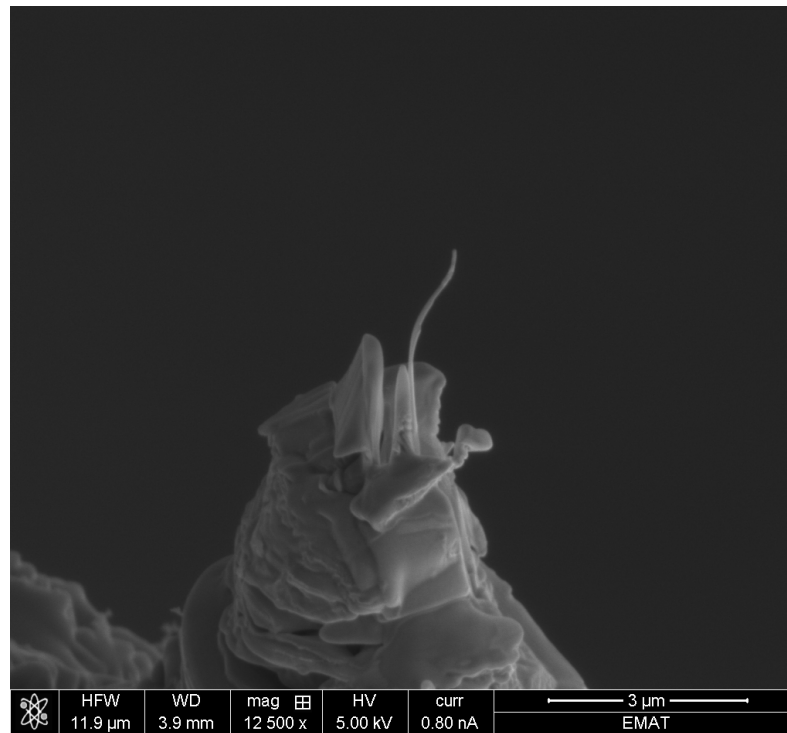
Mark targeted location with e-beam Pt



# Preparation of the needle



# Final prepared needle



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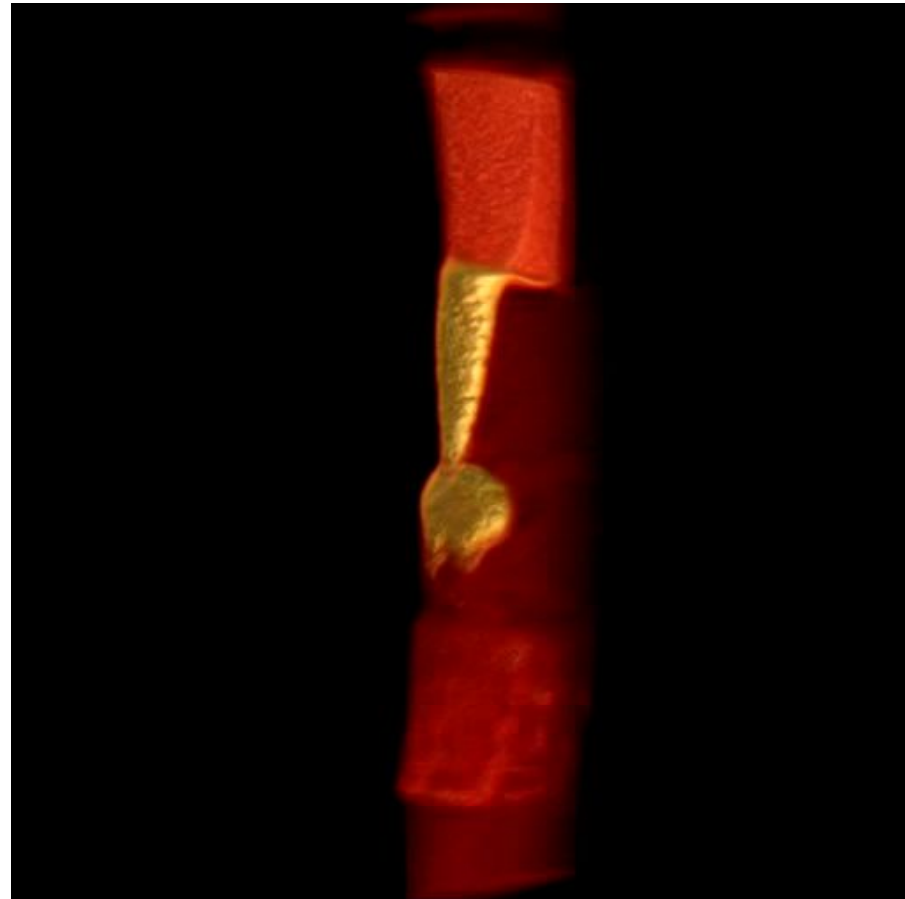
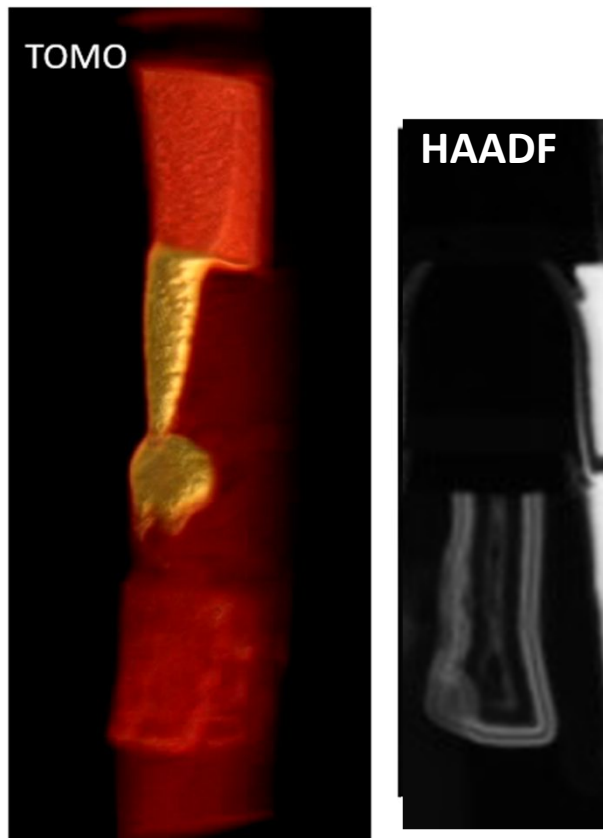
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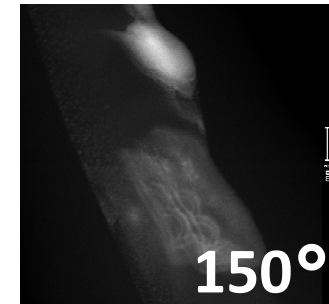
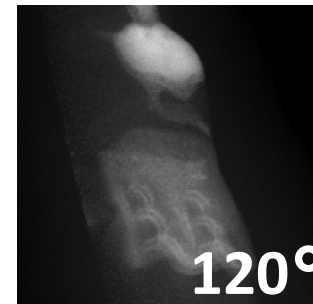
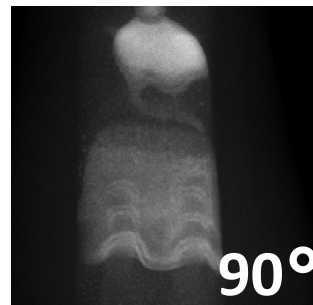
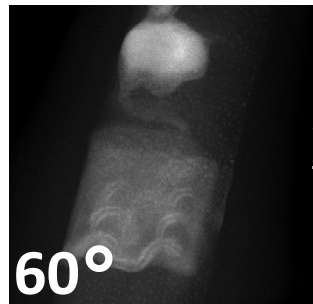
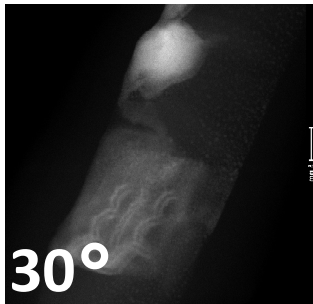
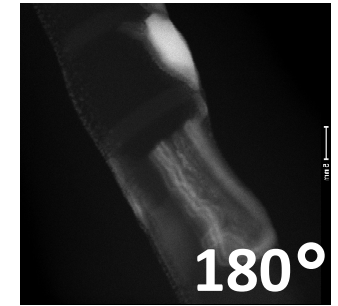
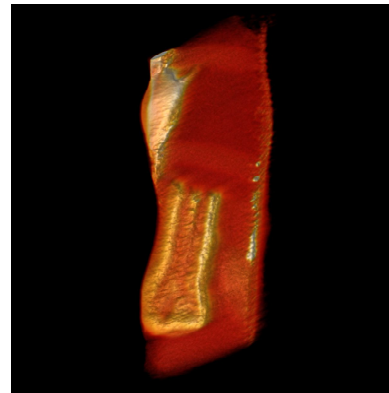
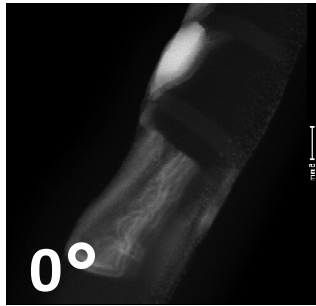
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# HAADF-tomography of the dummy gate

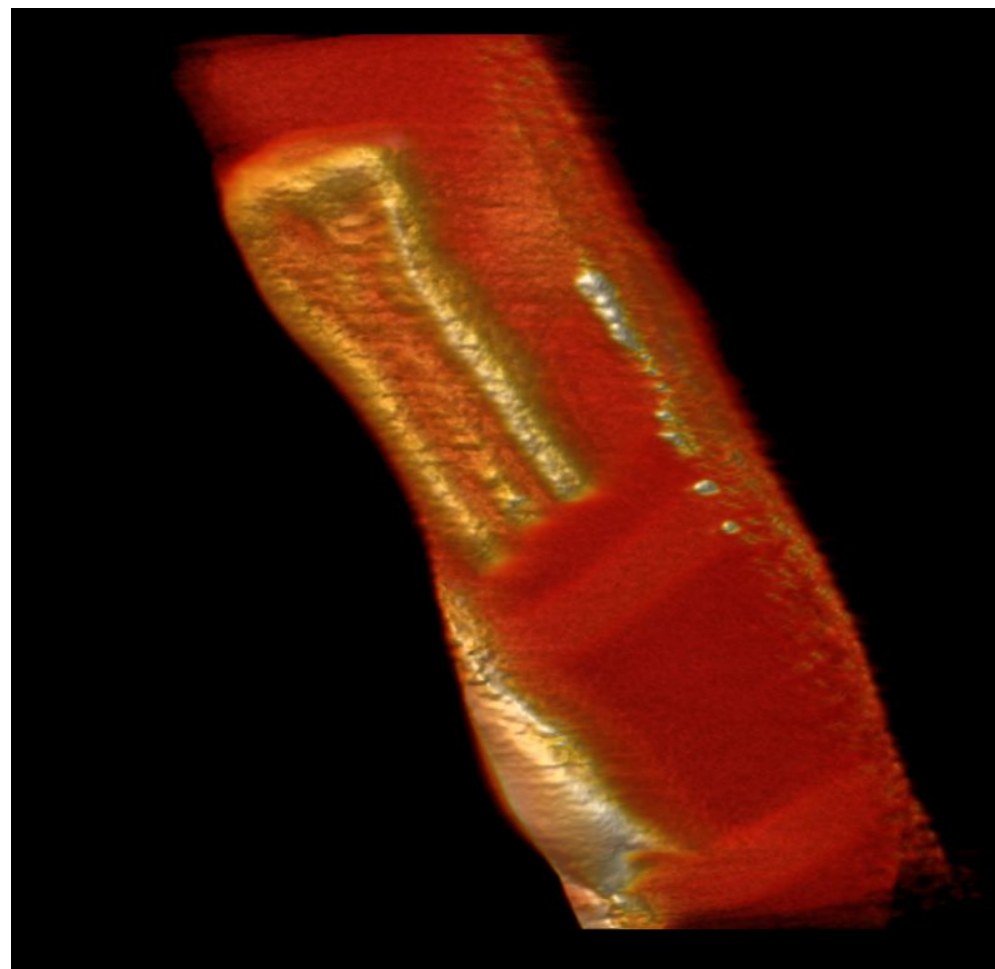
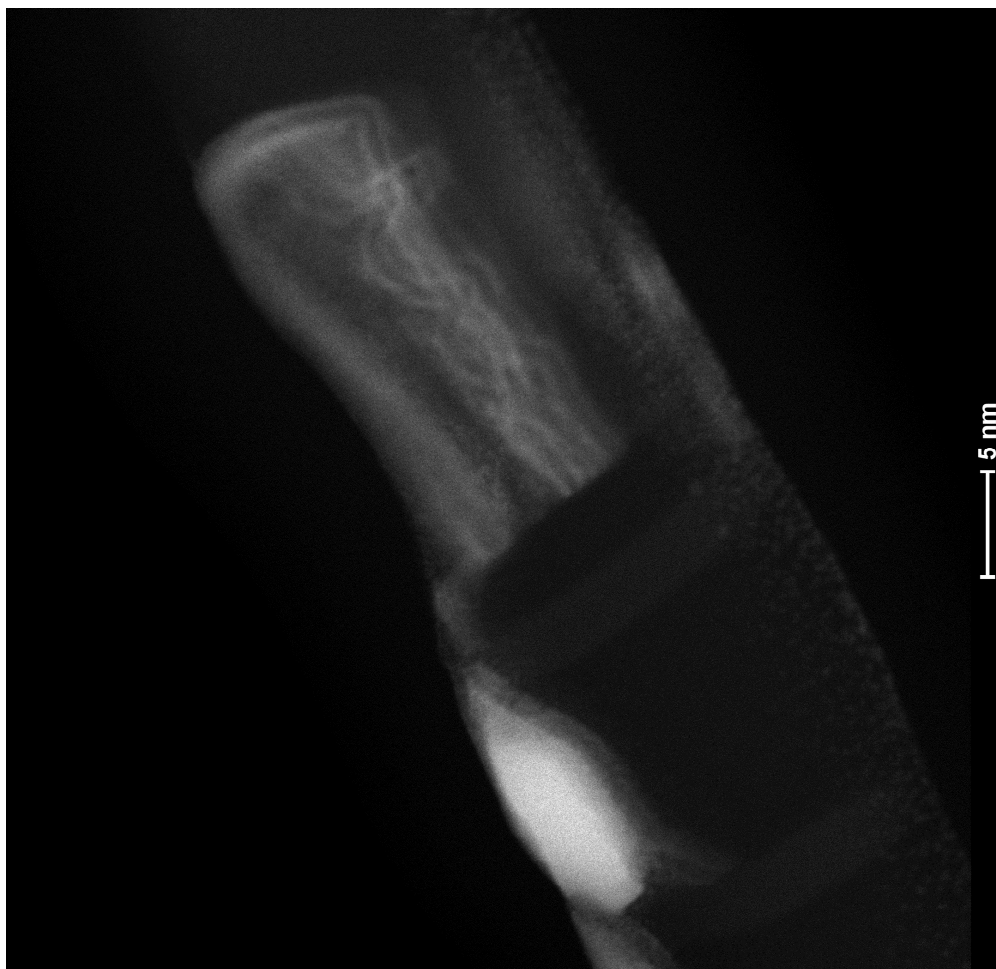




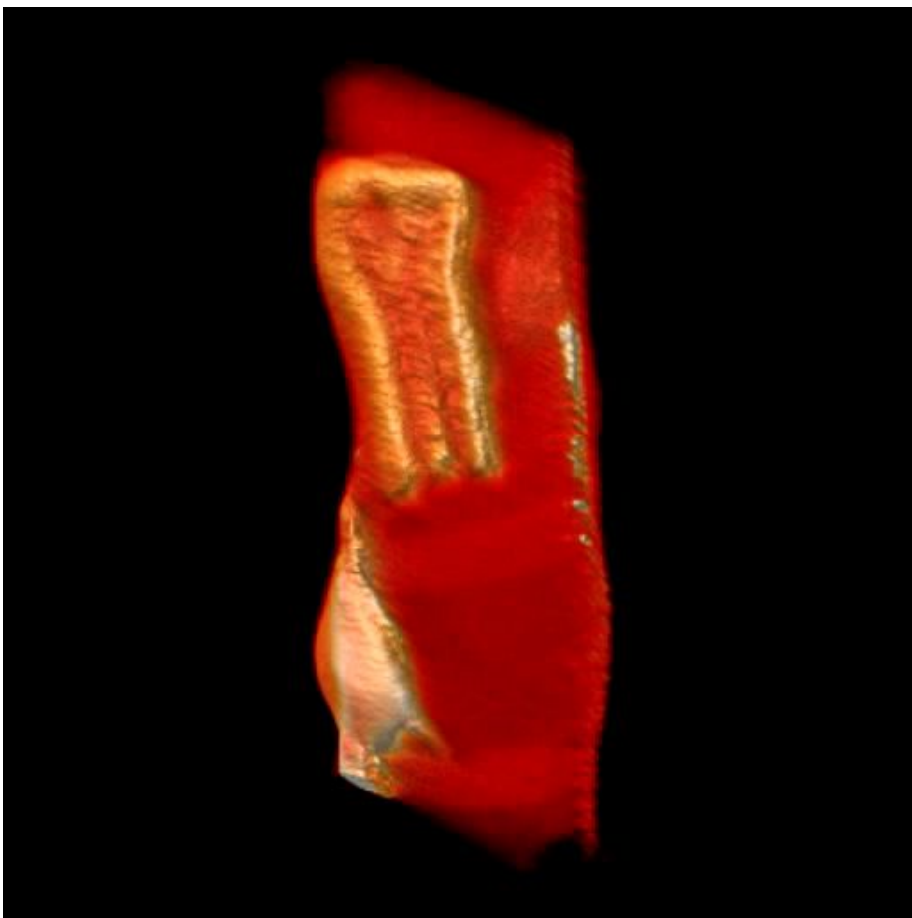
# HAADF → 5 Gb



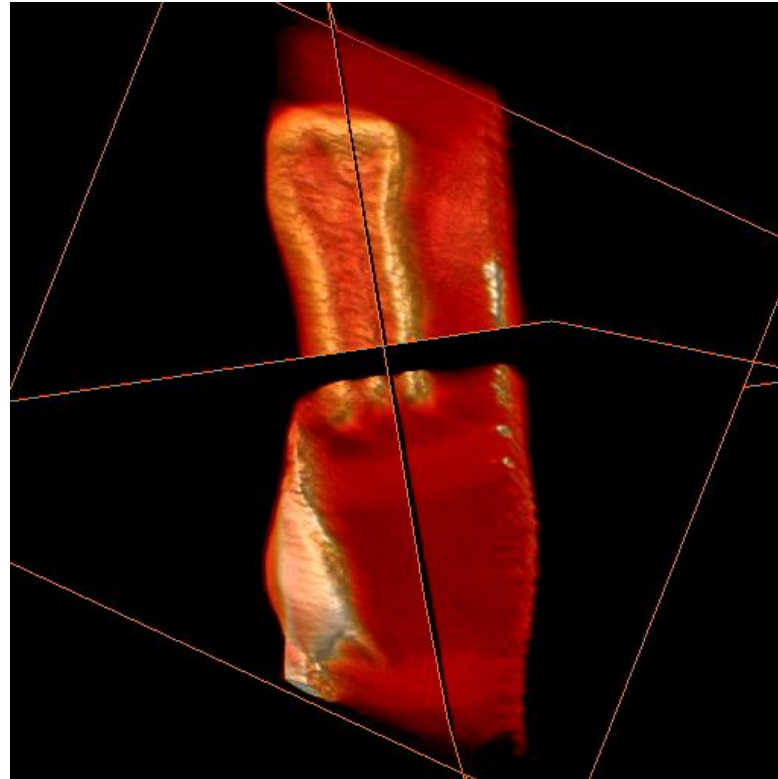
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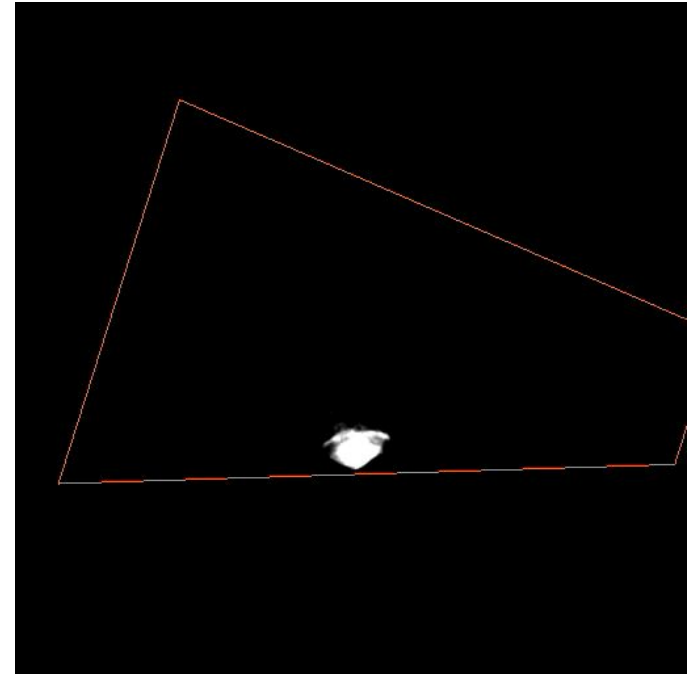
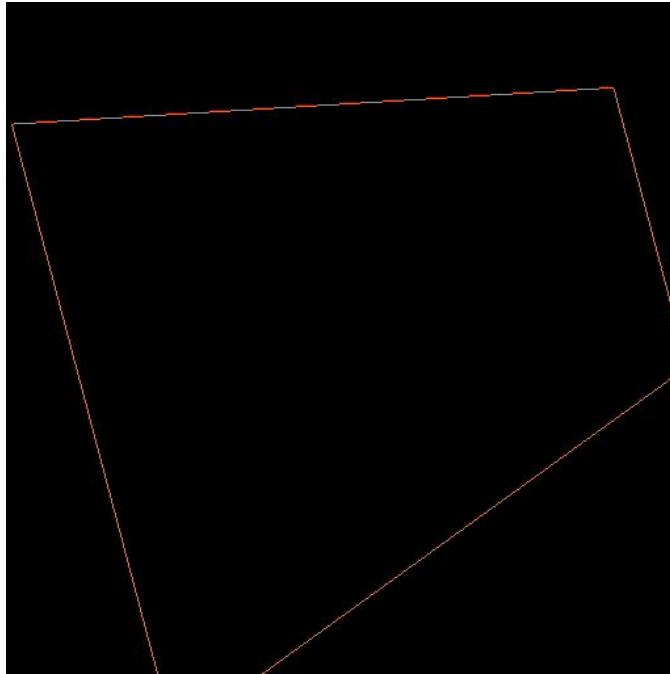
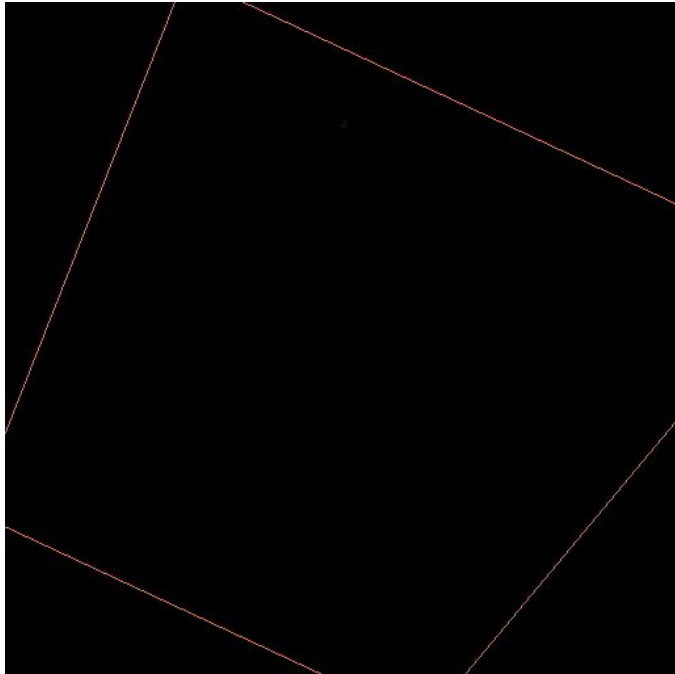


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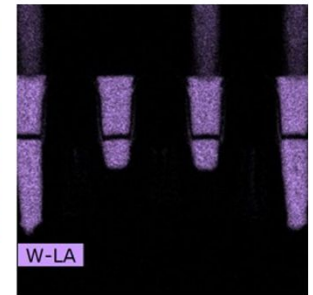
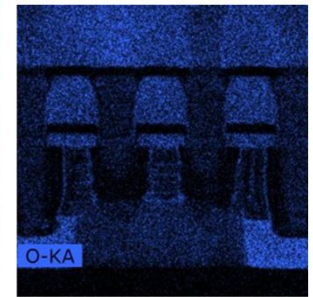
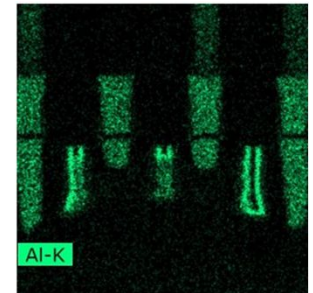
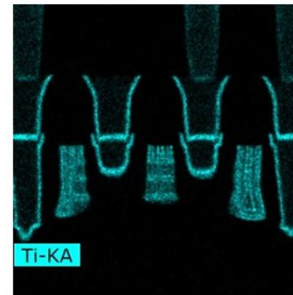
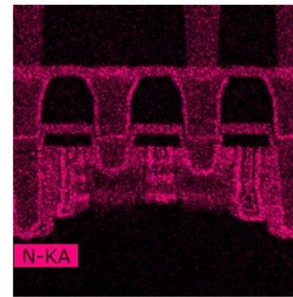
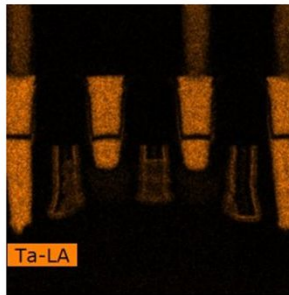
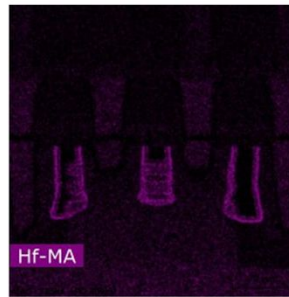
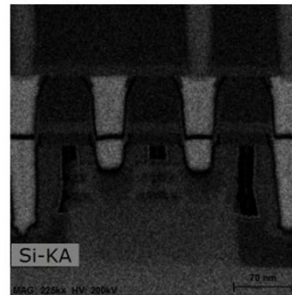
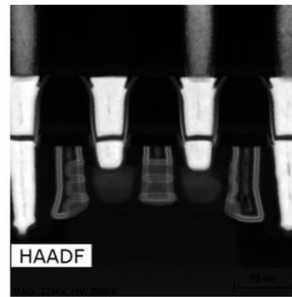
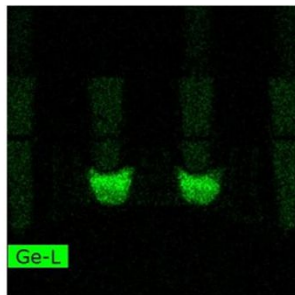
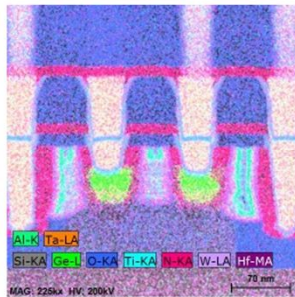
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# Lamella EDX

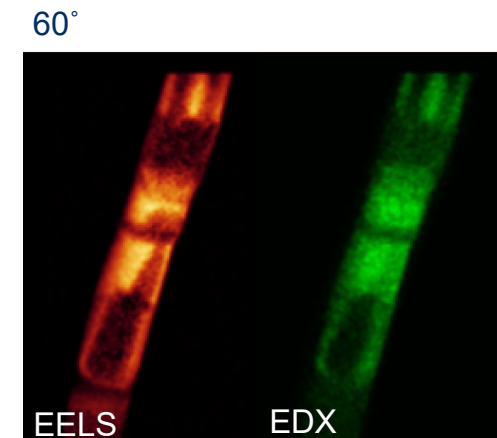
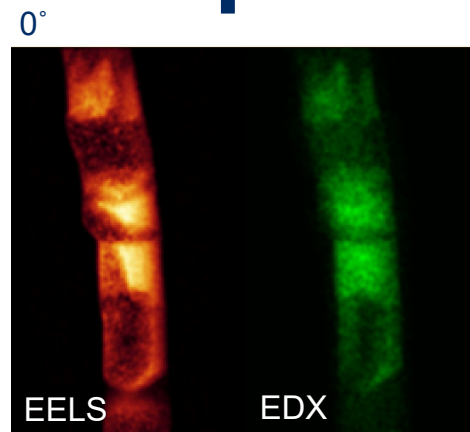
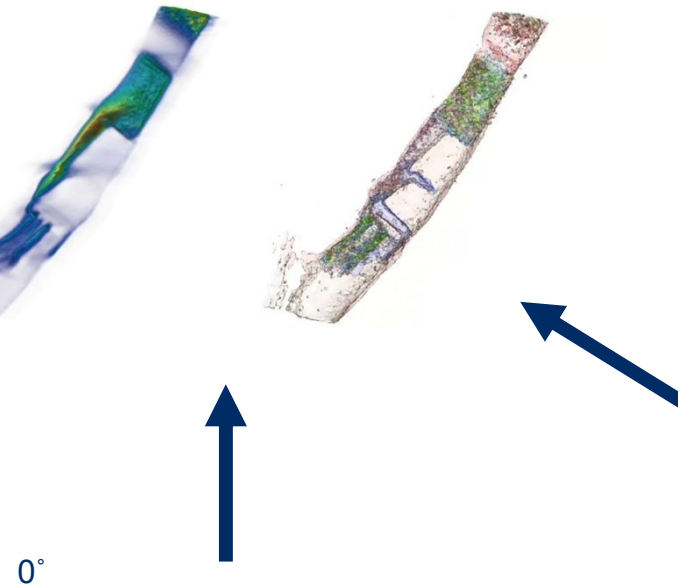
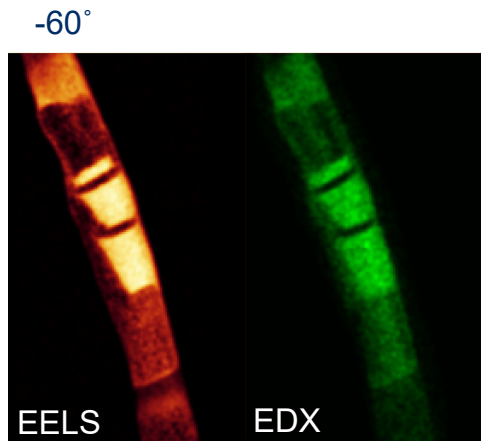
D05 PMOS - EDS01



## Methodology

# EELS+EDX → 60 Gb

EELS-EDX Tomography



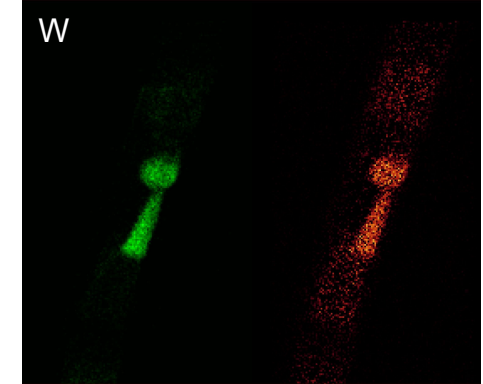
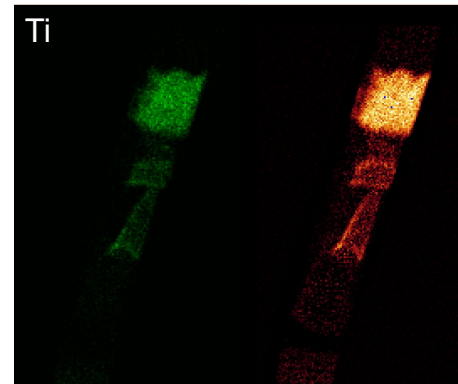
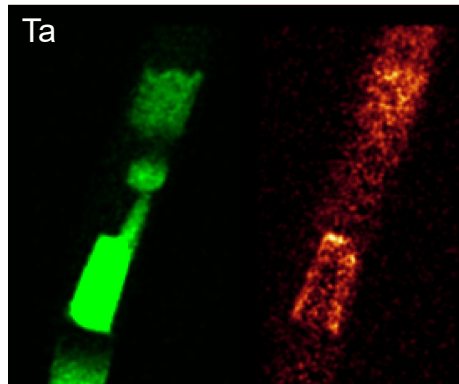
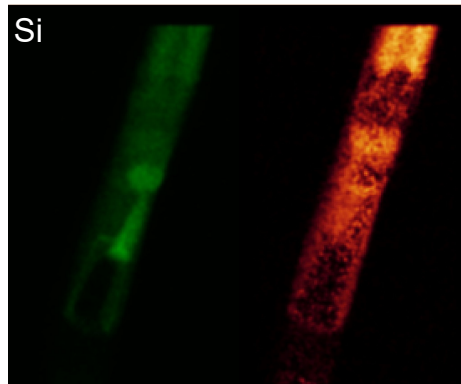
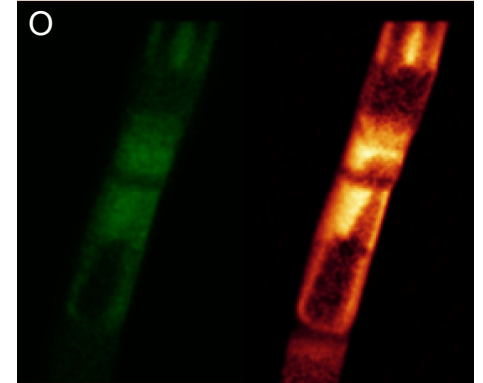
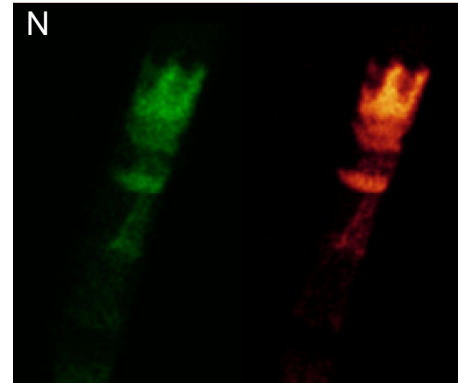
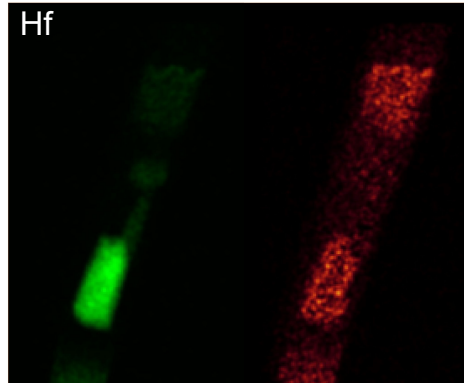
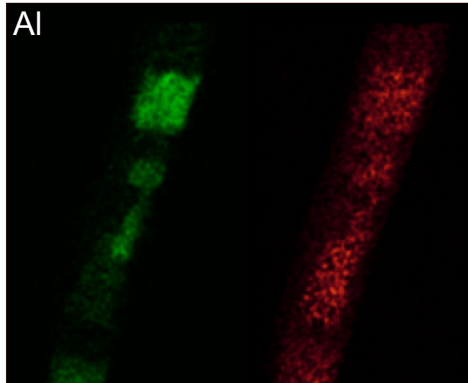
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# Signals

EDX

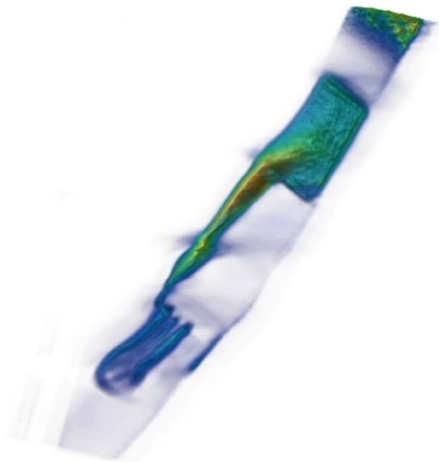
EELS



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# EELS-EDX Tomography

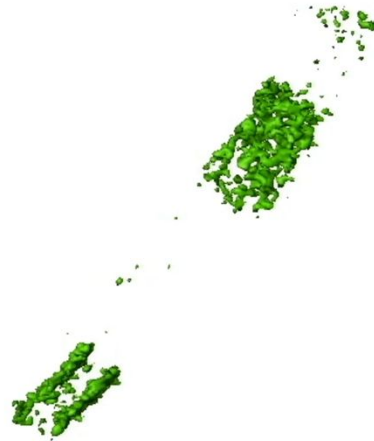
HAADF



EELS

Hf

Hf



EDX

Hf

Ge  
Hf

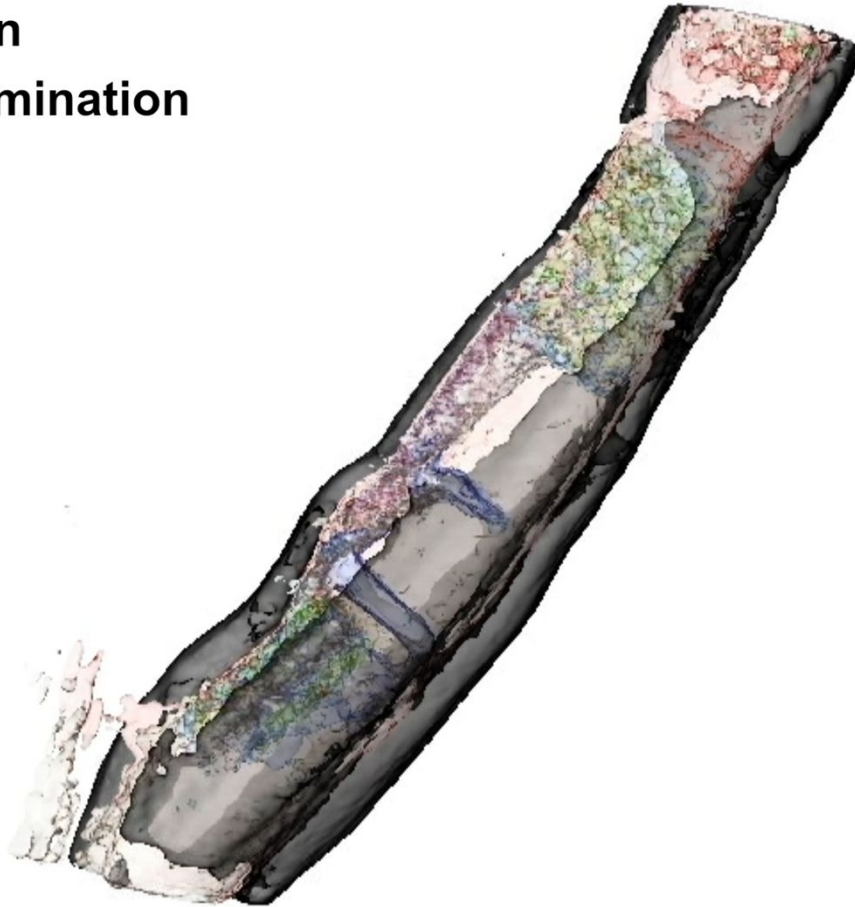


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# Carbon Contamination

## Carbon Contamination

C  
Hf  
Ta  
Ti  
Al  
W  
N  
Si  
O



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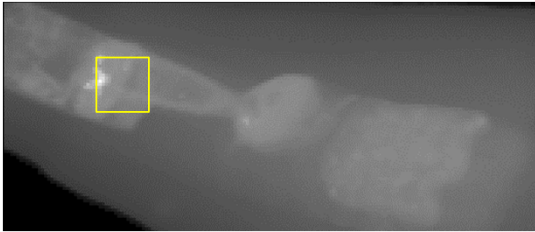
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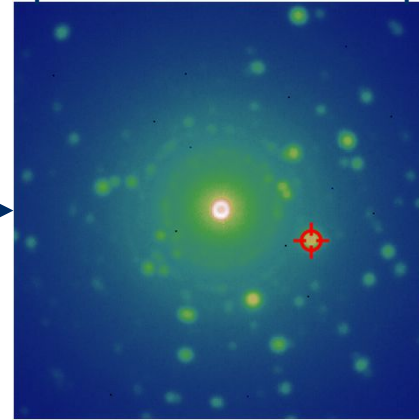
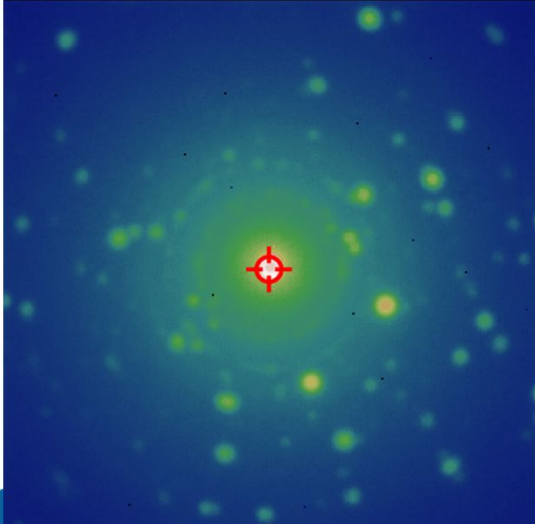
# 4DSTEM → 1 Tb

Map of a specific diffraction spot to identify the crystal phase on the sample

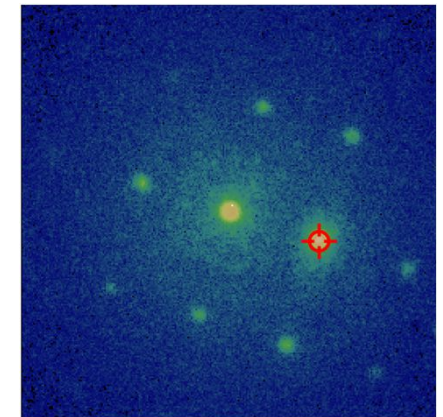
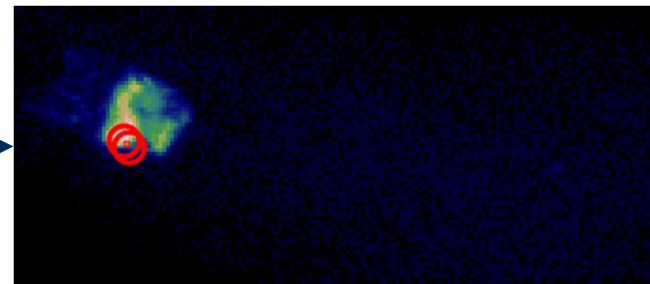
4DSTEM projection



Sum of diffraction patterns datasets



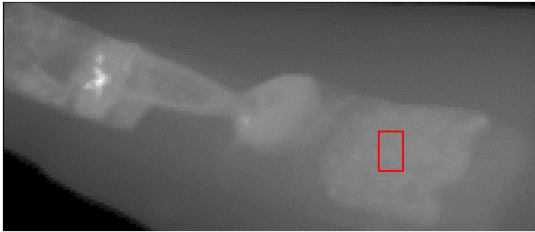
Study the crystal phase on individual diffraction patterns



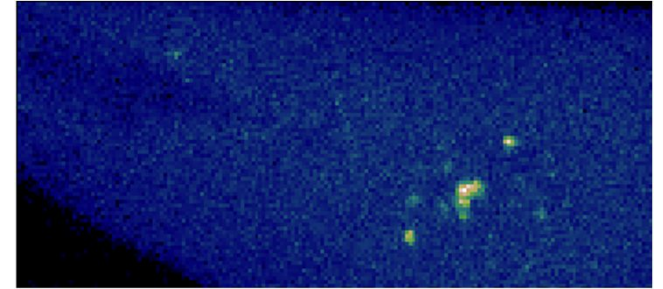
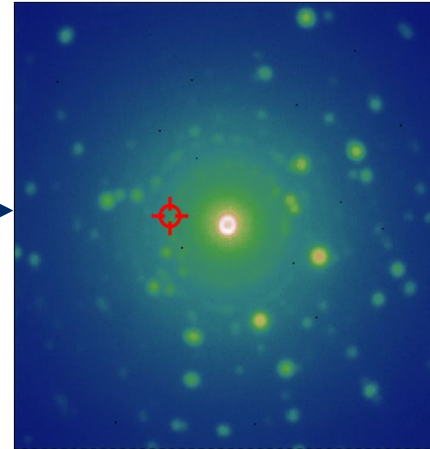
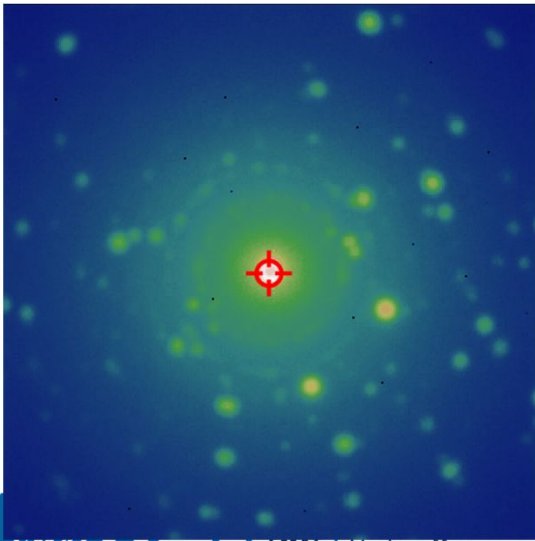
Cu <100>

Map of a specific diffraction spot to identify the crystal phase on the sample

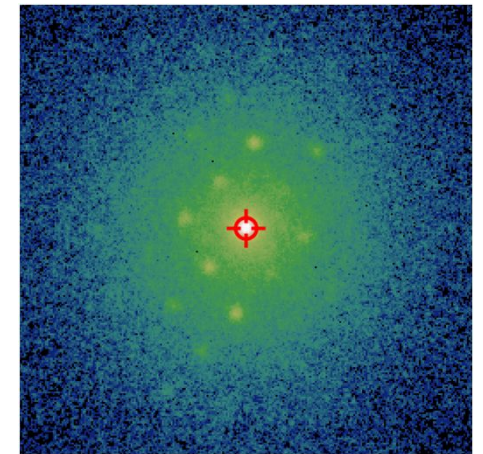
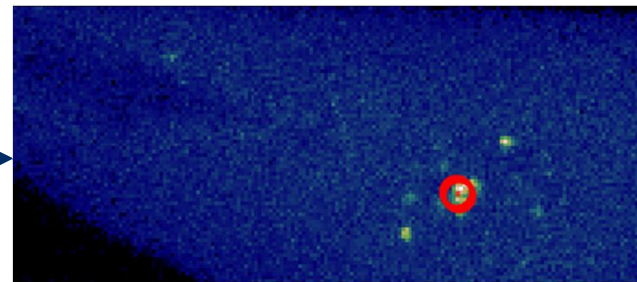
4DSTEM projection



Sum of diffraction patterns datasets



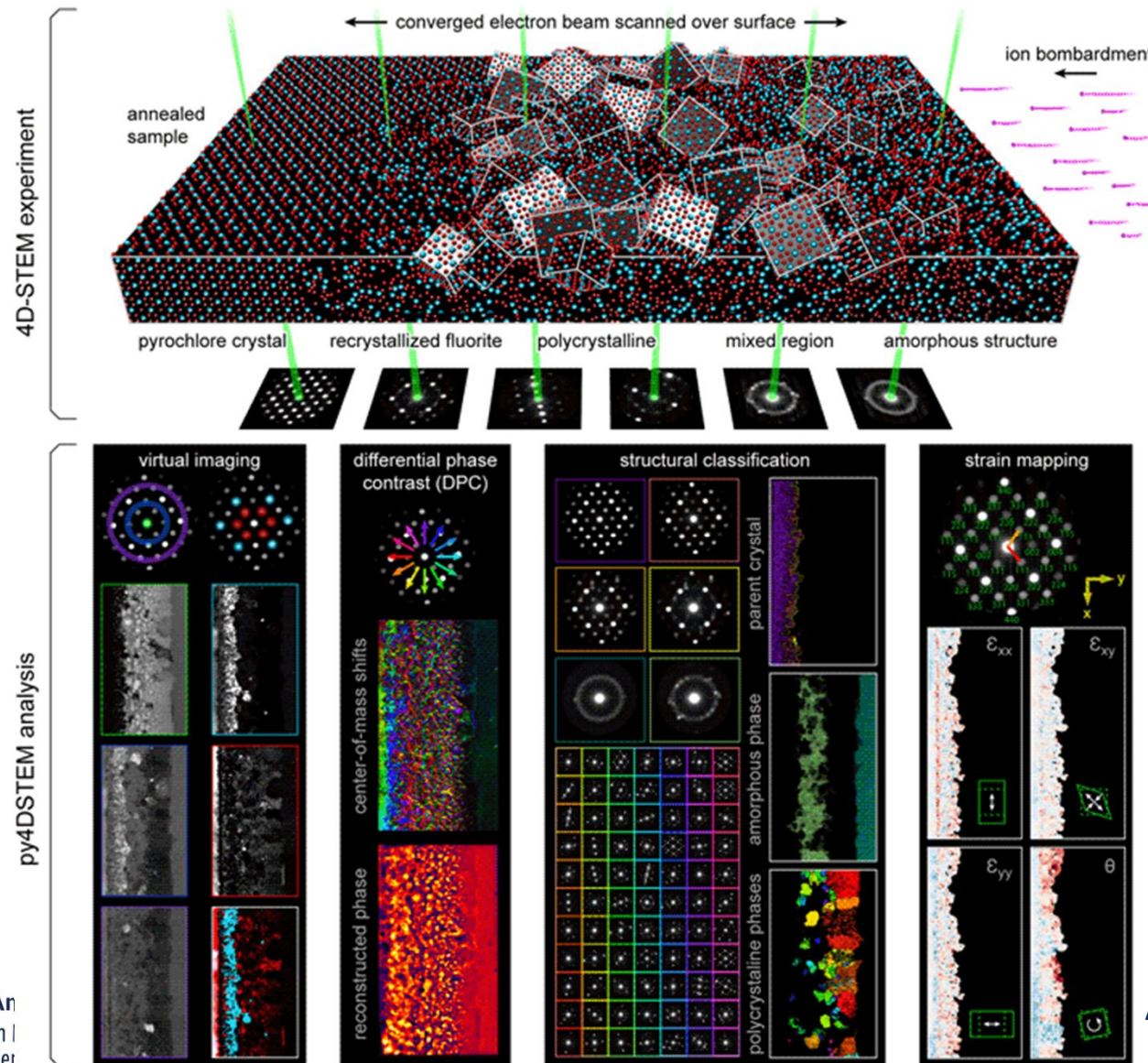
Study the crystal phase on individual diffraction patterns



Si <110>



# Possibilities in 4DSTEM



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# Outlook

- Tomography
  - became more accessible through fast tomo (faster) and developments in algorithms
  - Alternative SEEBIC available for shape determination
  - Multimodal
- 4DSTEM
  - Possible to determine stacking, defects, orientation of 2D materials
  - Possible to do in SEM
  - Can get information on crystalline and amorphous materials
- Tomography of complex semiconductor
  - Combination of different techniques possible and necessary
  - Sample preparation is the most critical step
  - Necessity to have 3D information



# Acknowledgements





***Thank you for your attention***

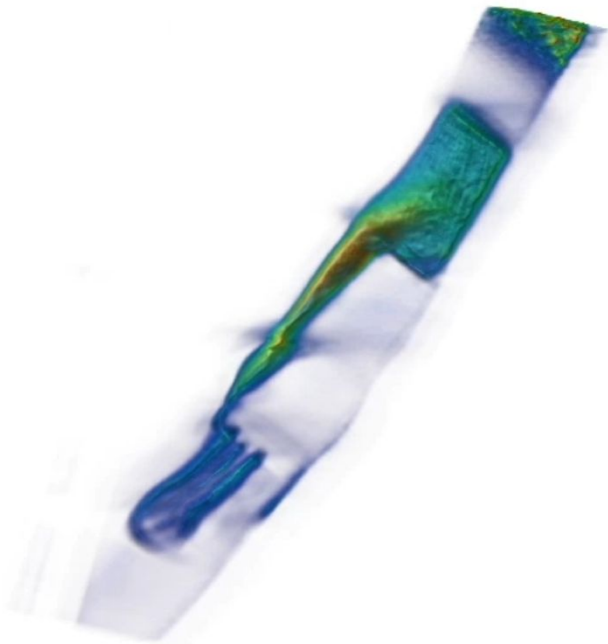
**QUESTIONS ???**



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# EELS Tomography

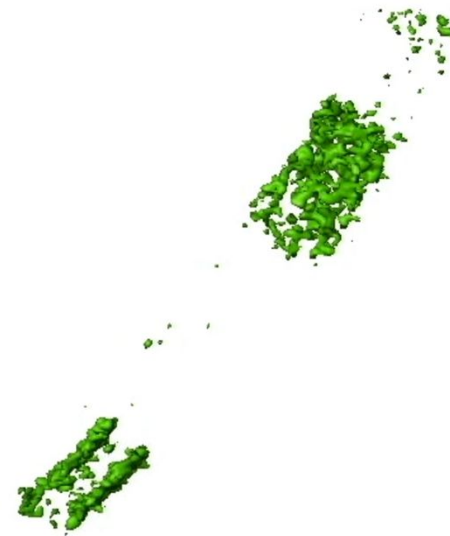
HAADF



EELS

Hf

Hf

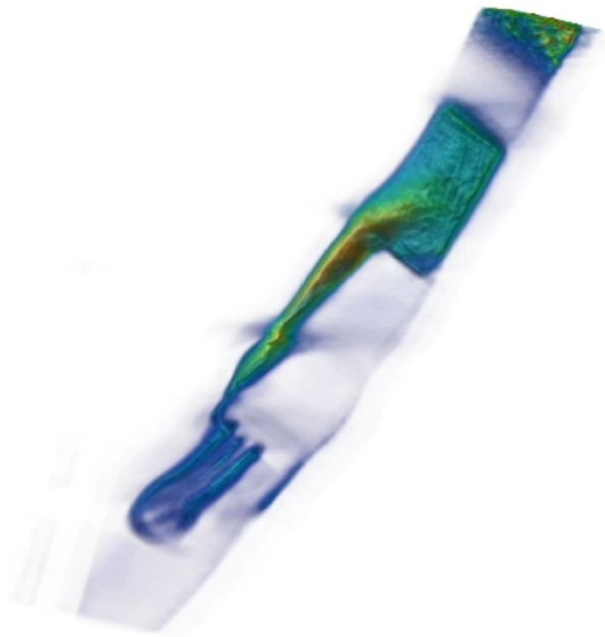


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# EDX Tomography

HAADF



EDX

**Ge**

Ge



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